



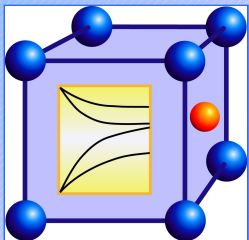
Polish Academy of Sciences



**Institute of Nuclear Physics
Polish Academy of Sciences
Cracow, Poland**

Material properties studied with PHONON

Krzysztof PARLINSKI



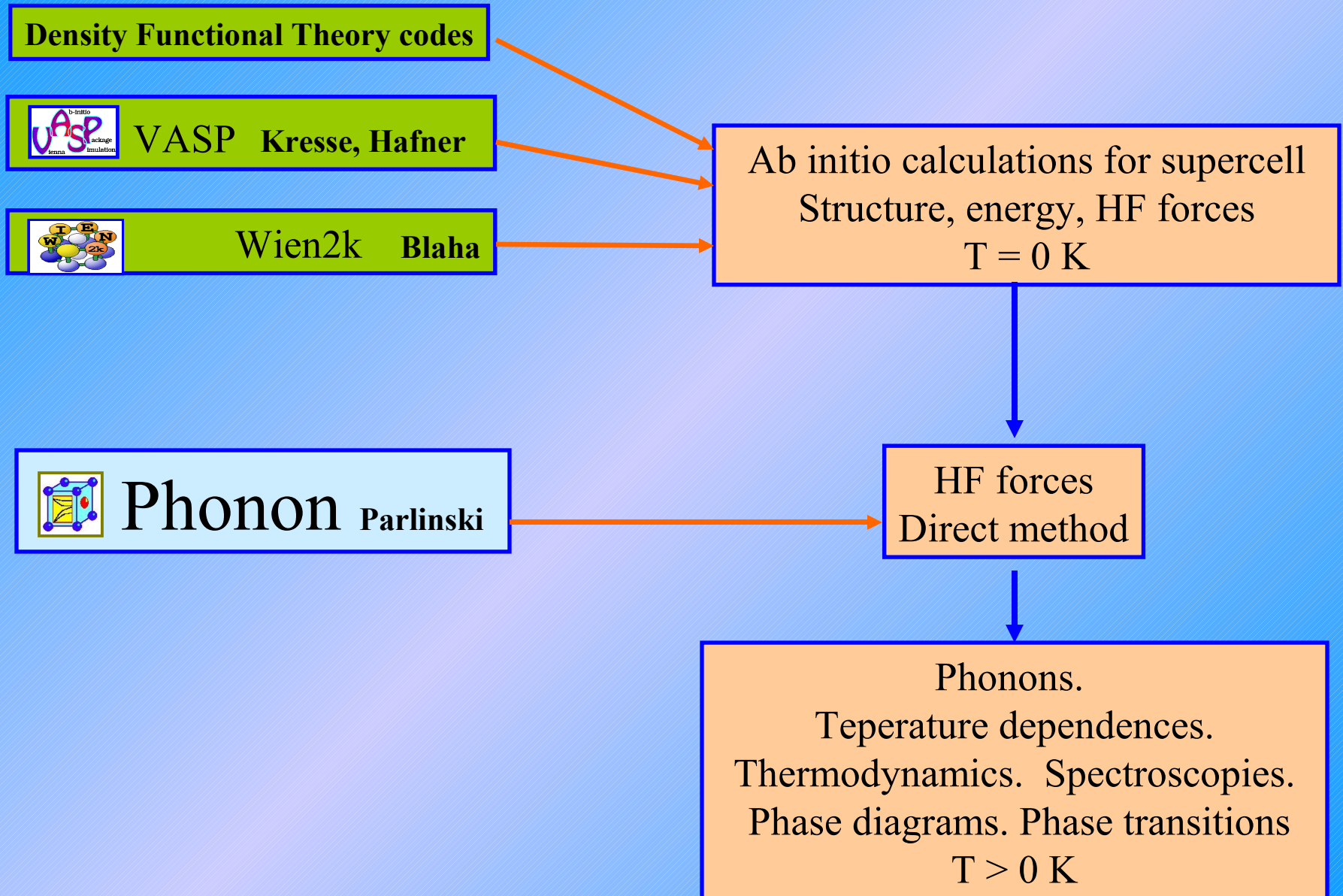
Computing for Materials



**Institute of Technics
Pedagogical University
Cracow, Poland**

Temperature of materials is
determined (mainly) by phonons

Strategy of computation



Some quantities to be calculated using DFT, and DFT + Phonon

DFT T=0 K, any P

- *Lattice parameters
- *Atomic positions
- *Crystal structure
- External stresses and pressure
- *HF forces acting on atoms
- Charge distribution $\rho(x,y,z)$
- Magnetic moments
- Electronic states, electronic gap
- Electronic structure
- Electronic density of states
- *Polarization
- etc.
- * - needed for phonon calculations

Systems:

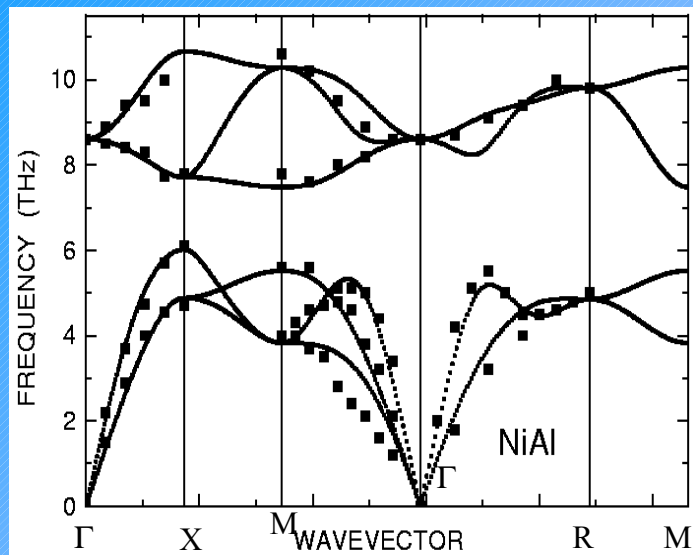
- Crystals (230 space groups)
- Surfaces (on slab)
- Multilayers, interfaces
- (Point) defects, vacancies
- Amorphous
- etc. All systems represented by SUPERCELL

DFT+Phonon T>0 K, any P

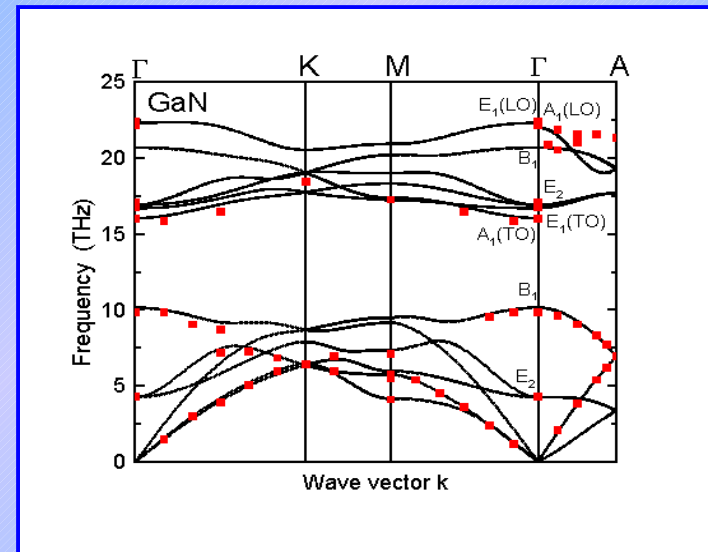
- Phonon dispersion relations $\omega(\mathbf{k},j)$
- Phonon intensities (different filters)
- LO/TO splitting from $Z^*(m)$ and ϵ_{inf}
- Irreducible representations of modes
- Phonon intensities in many Brillouin Zones
- Gruneissen parameters
- Atomic displacements involved in a mode
- Animate phonon(s) motion
- Find electronic state coupled to $\omega(\mathbf{k},j)$
- Phonon density of states $g(\omega)$
- Partial phonon density of states $g_{\mu,i}(\omega)$
- Thermodynamical functions: E, S, F, G, c_v
- Debye-Waller factor $\langle u^2(\mu) \rangle$
- Thermal expansion
- Neutron & x-rays inelastic coherent scattering
- Neutron inelastic incoherent scattering
- Nuclear inelastic scattering (NIS) with x-rays
- Searching for soft modes & phase transitions
- Construct (T,P) - phase diagram
- Find chemical reactions constant
- Diffusion constant
- Heat conductivity (?)

Comparing computed phonon dispersion curves with experiment

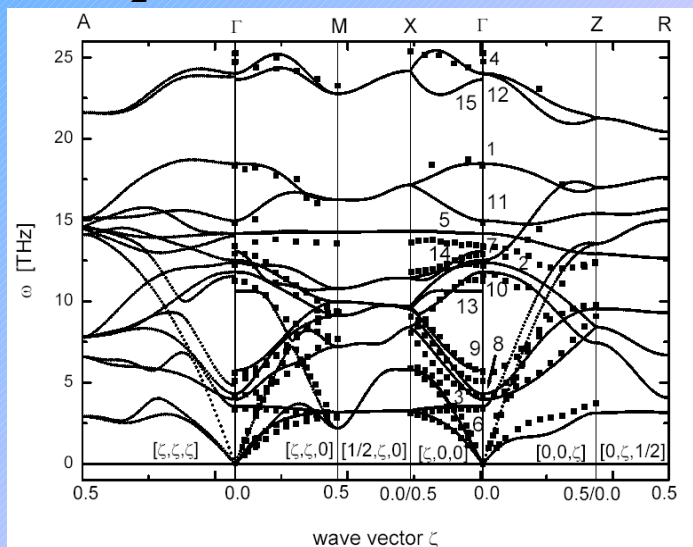
NiAl Cubic primitive



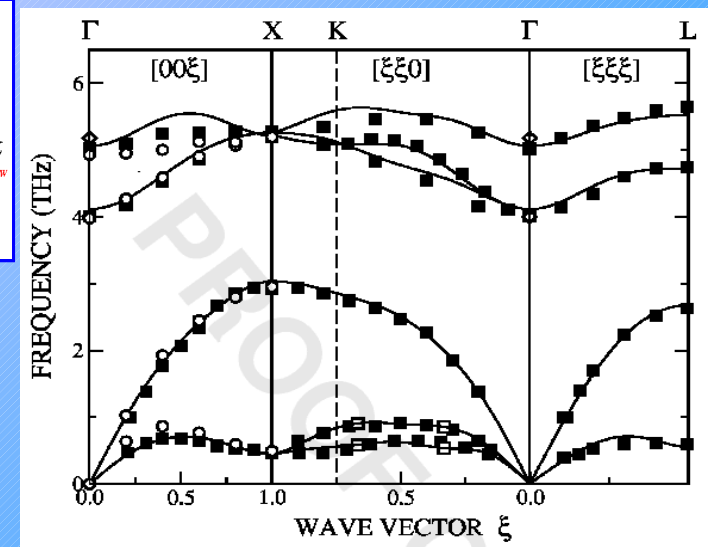
GaN Hexagonal wurzite



TiO₂ Tetragonal



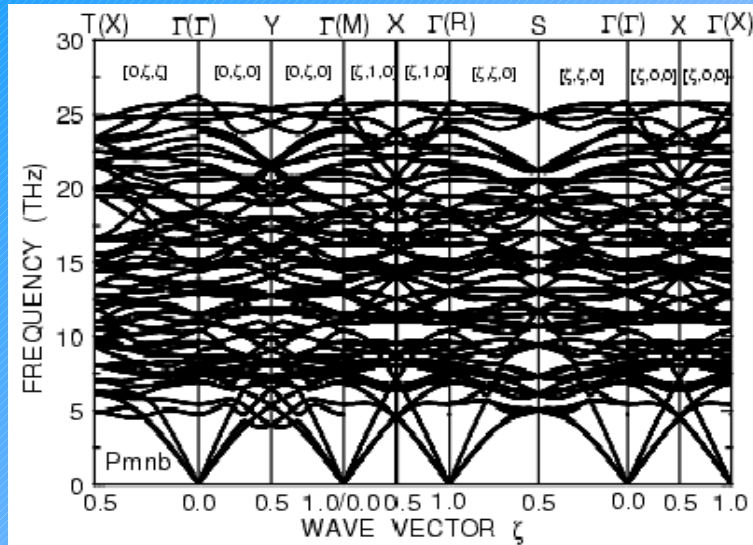
HgSe Cubic face centered



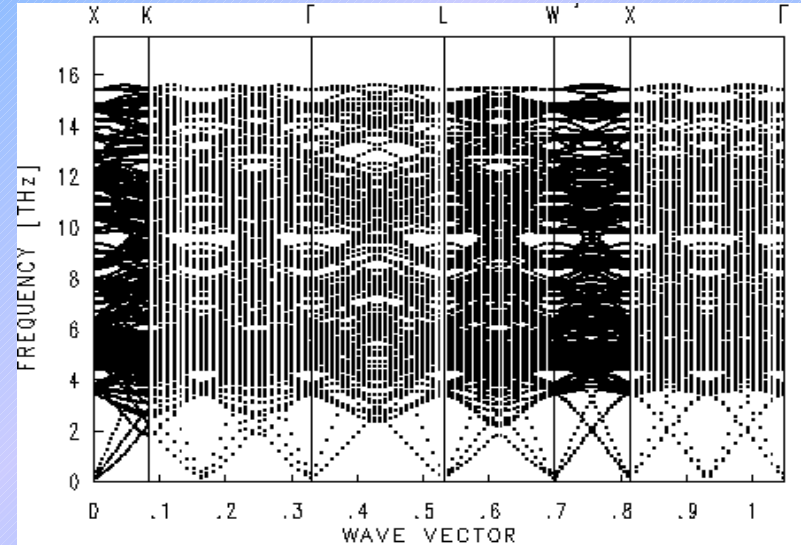
Mercury selenate

Phonon dispersion curves of more complex systems

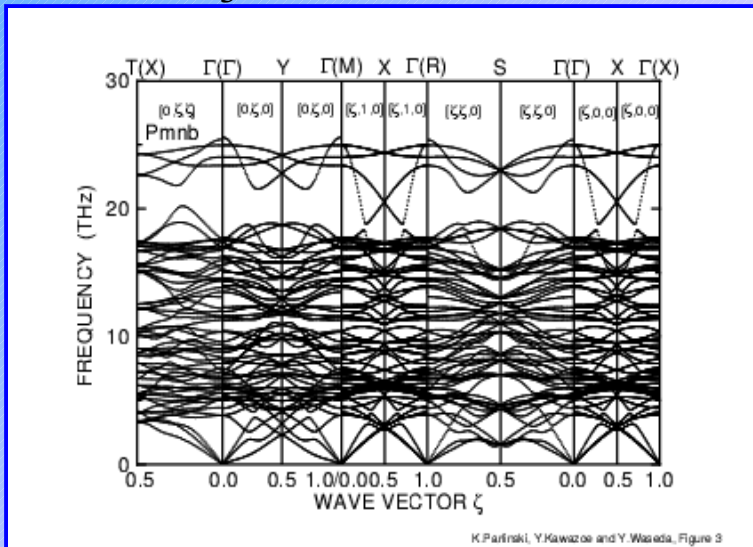
MgSiO₃ orthorhombic



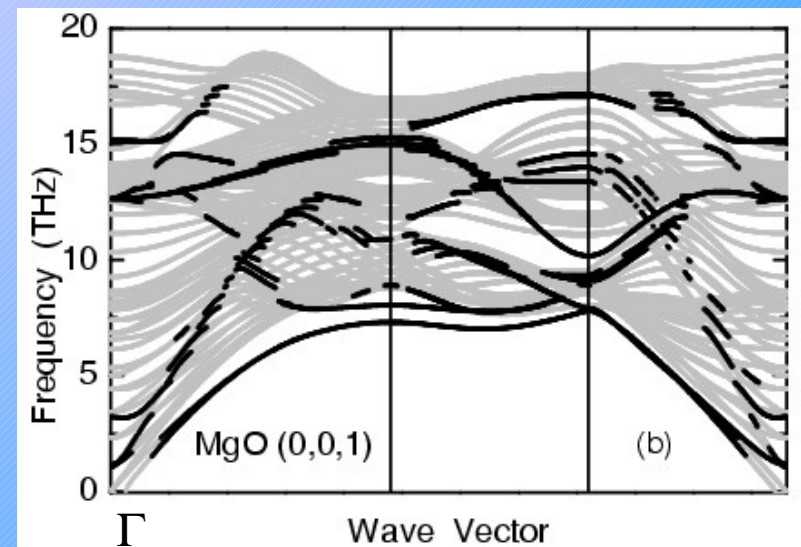
Cubic CoO + one vacancy



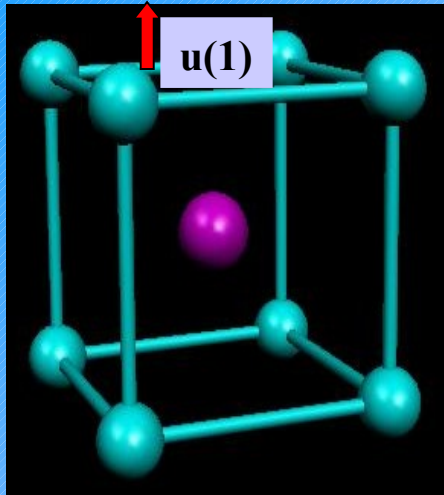
CaTiO₃ orthorhombic



Surface MgO(001)_M X Γ



$u(1)$



Potential: $V = \frac{1}{2} \sum_{n,m} \Phi(n, m) u(n) u(m)$

Force:

$$F(n) = - \sum_m \Phi(n, m) u(m)$$

$\Phi(n, m)$ – force constant - matrix 3x3
between atom n and m

Dynamical matrix:

$$D(k) = 1/M \sum_m \Phi(0, m) \exp[-ik(R(0)-R(m))]$$

Phonons:

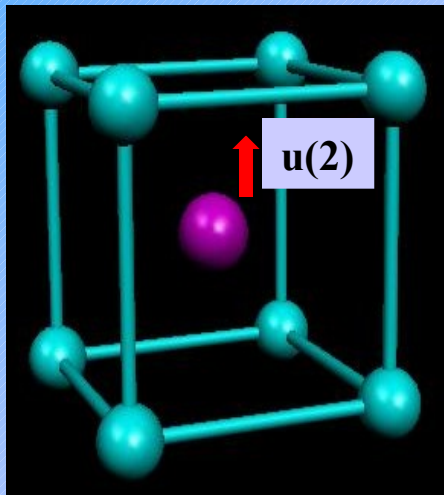
$$\omega^2(k) e(k) = D(k) e(k)$$

$e(k)$ – polarization vector, eigenvector

Hellmann-Feynman forces $F(n)$ arise due
to displacements of an atom in supercell

$$F(n) \longrightarrow \Phi(n, m) \longrightarrow \omega(k)$$

$$\text{Potential: } V = \frac{1}{2} \sum_k \omega^2(k) |Q(k)|^2$$

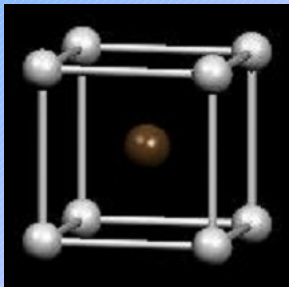
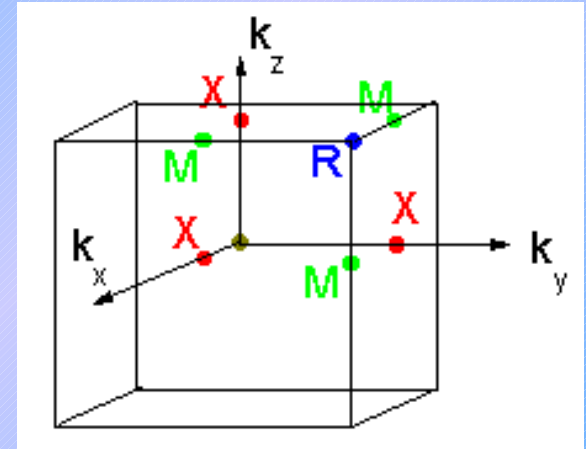


SUPERCCELL

Exact wave vectors

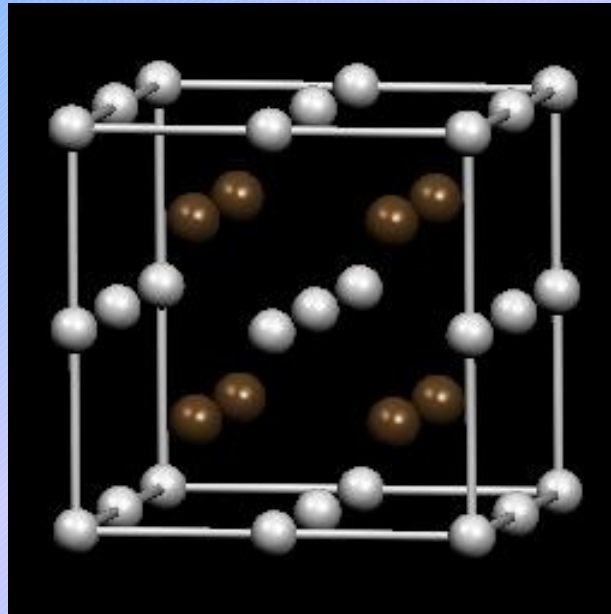
Wave vectors, which are commensurate with the size of supercell, provide phonon frequencies being „exact”, independent on the interaction range.

Brillouin zone for simple cubic



Supercell 1x1x1

Exact points Γ



2x2x2

Γ , X, M, R

Procedure

- Crystals are always treated as **supercell**. Supercell is a fragment of crystal being the multiplication of primitive unit cells. On supercell are imposed **3d** periodic boundary conditions.
- Optimization of crystal supercell by *ab initio* program (VASP, Wien2k)
- Calculation of Hellmann-Feynman atomic forces $\mathbf{F}$, created by single atom displacement
- Use of direct method to get from $\mathbf{F}$ (VASP, Wien2k), force constants Φ (Phonon)
- Calculation of dynamical matrix and phonon frequencies

Some results of calculations

Phonons in bulk crystals.

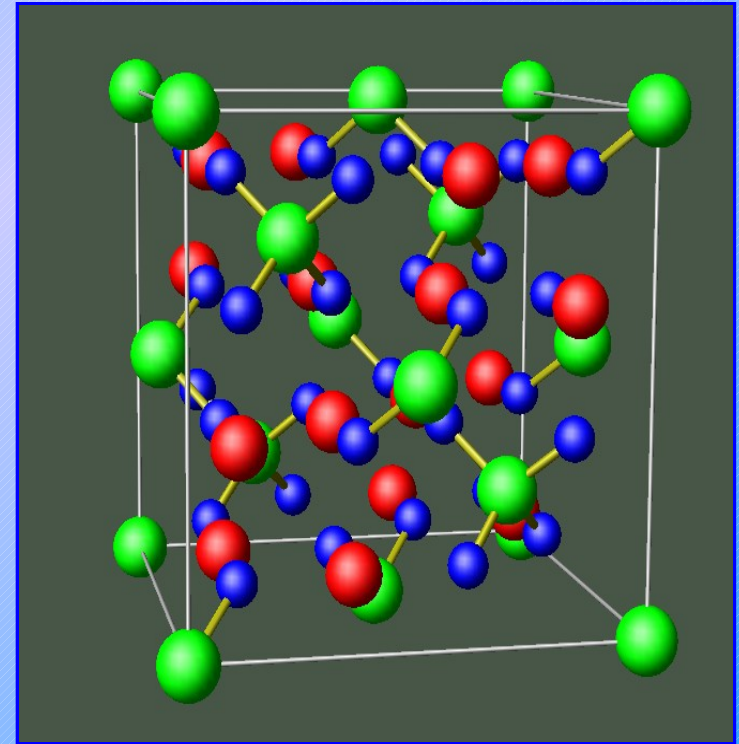
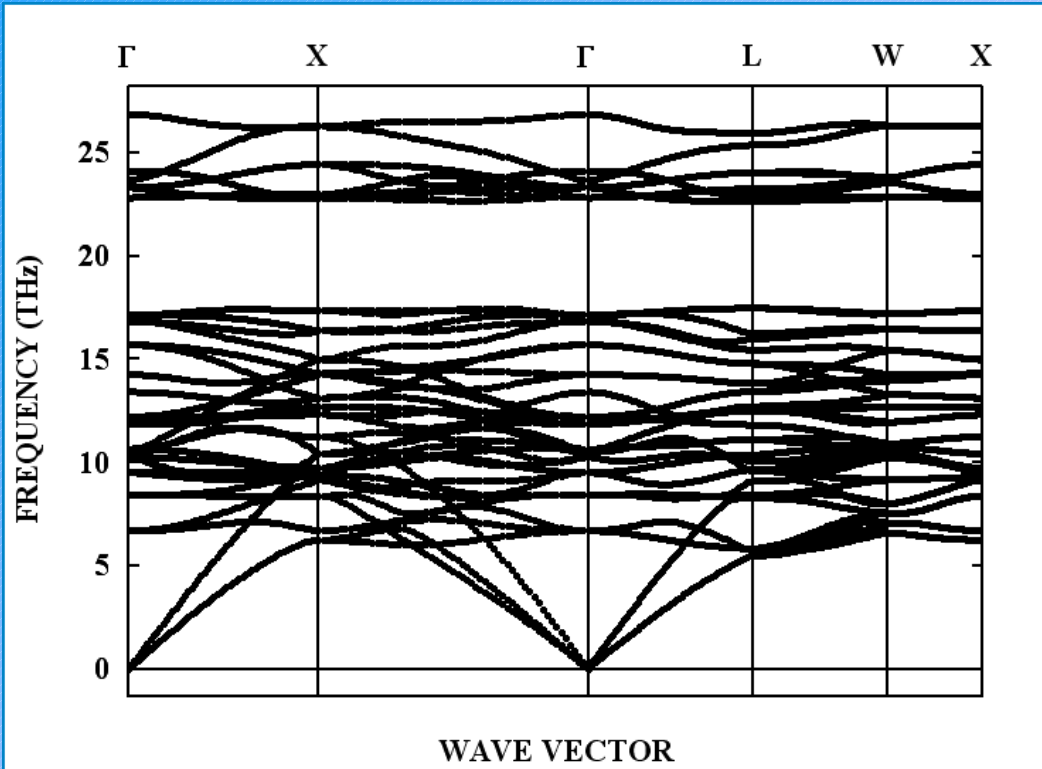
Example: Mg_2SiO_4



Cubic, Fd-3m,

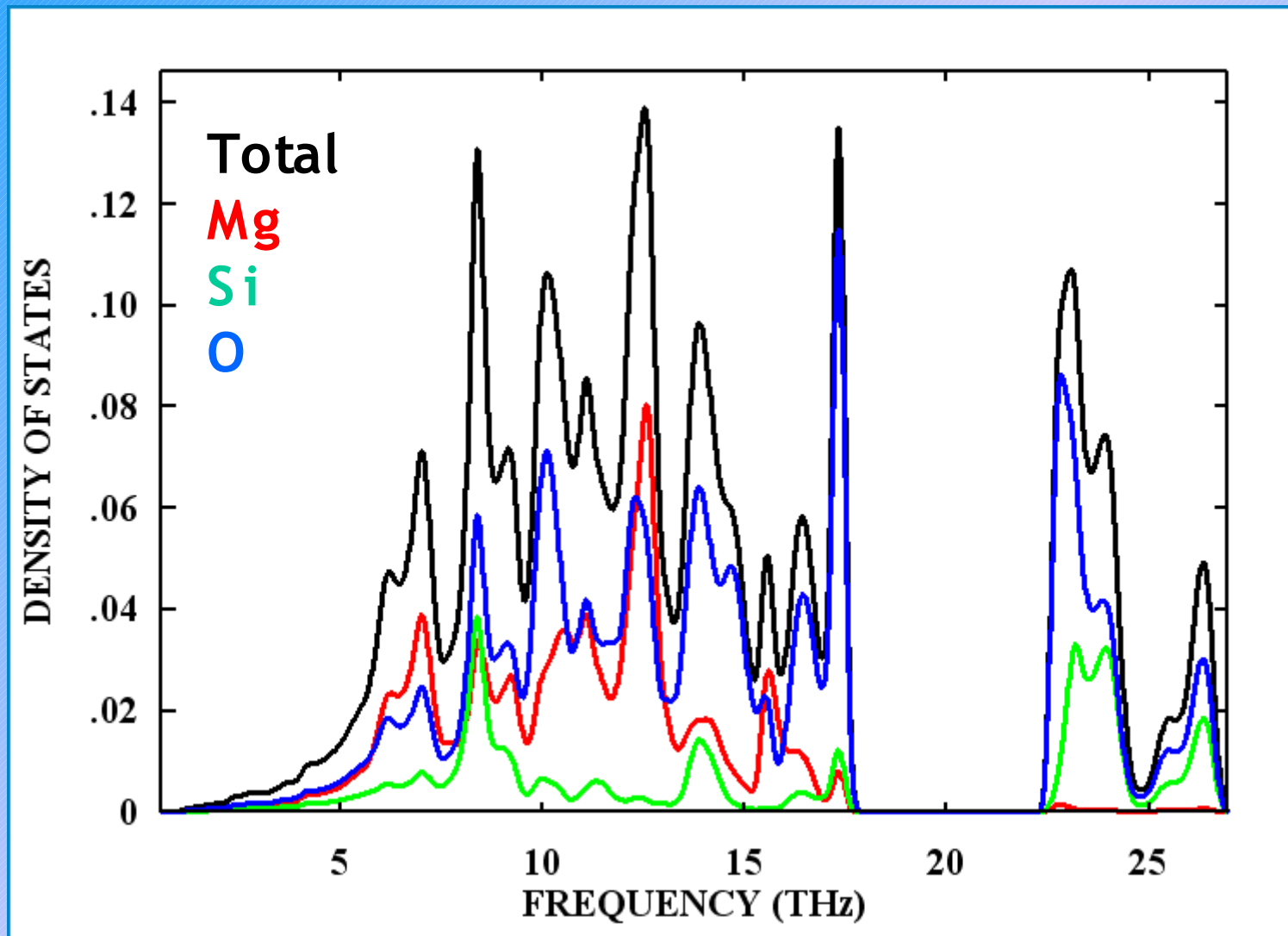
Phonon dispersion relations

7 atoms x 2 x 3 = 42 phonon dispersion curves



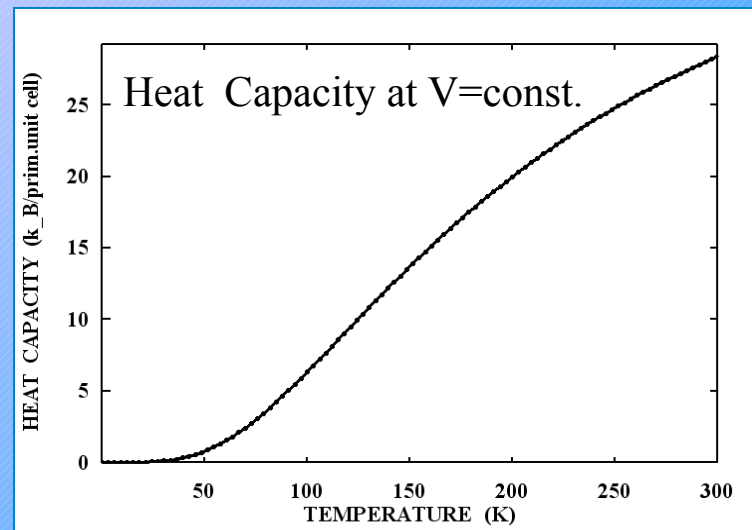
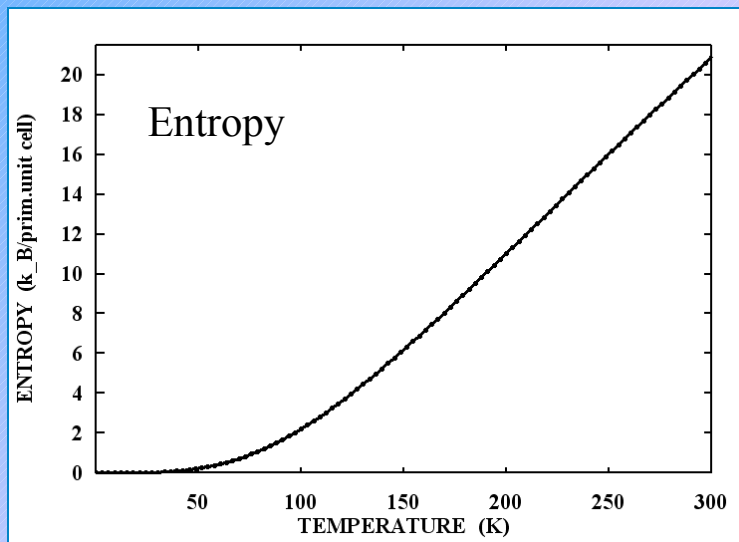
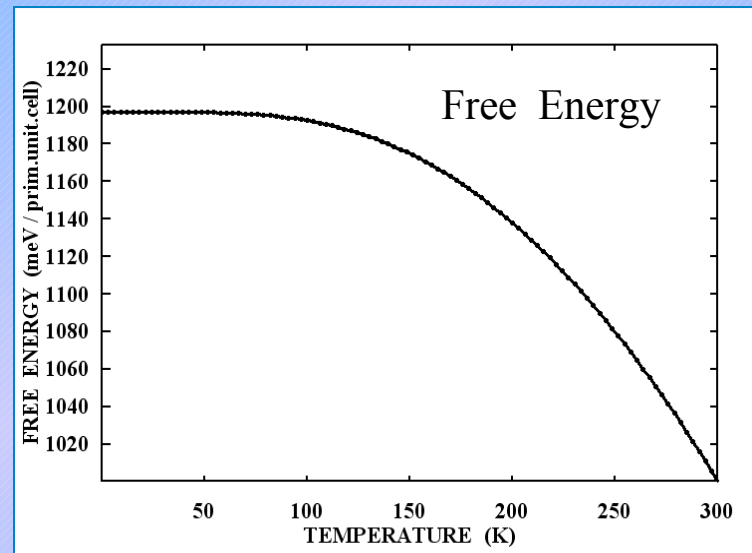
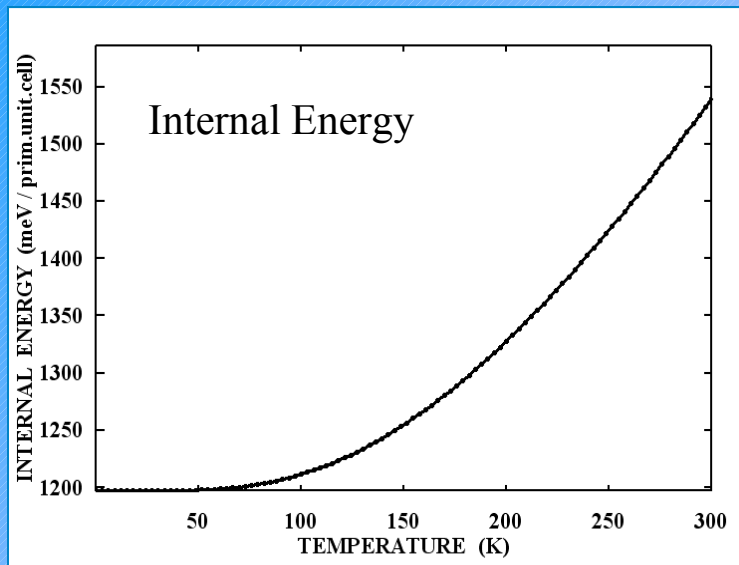


Density of States (DOS)



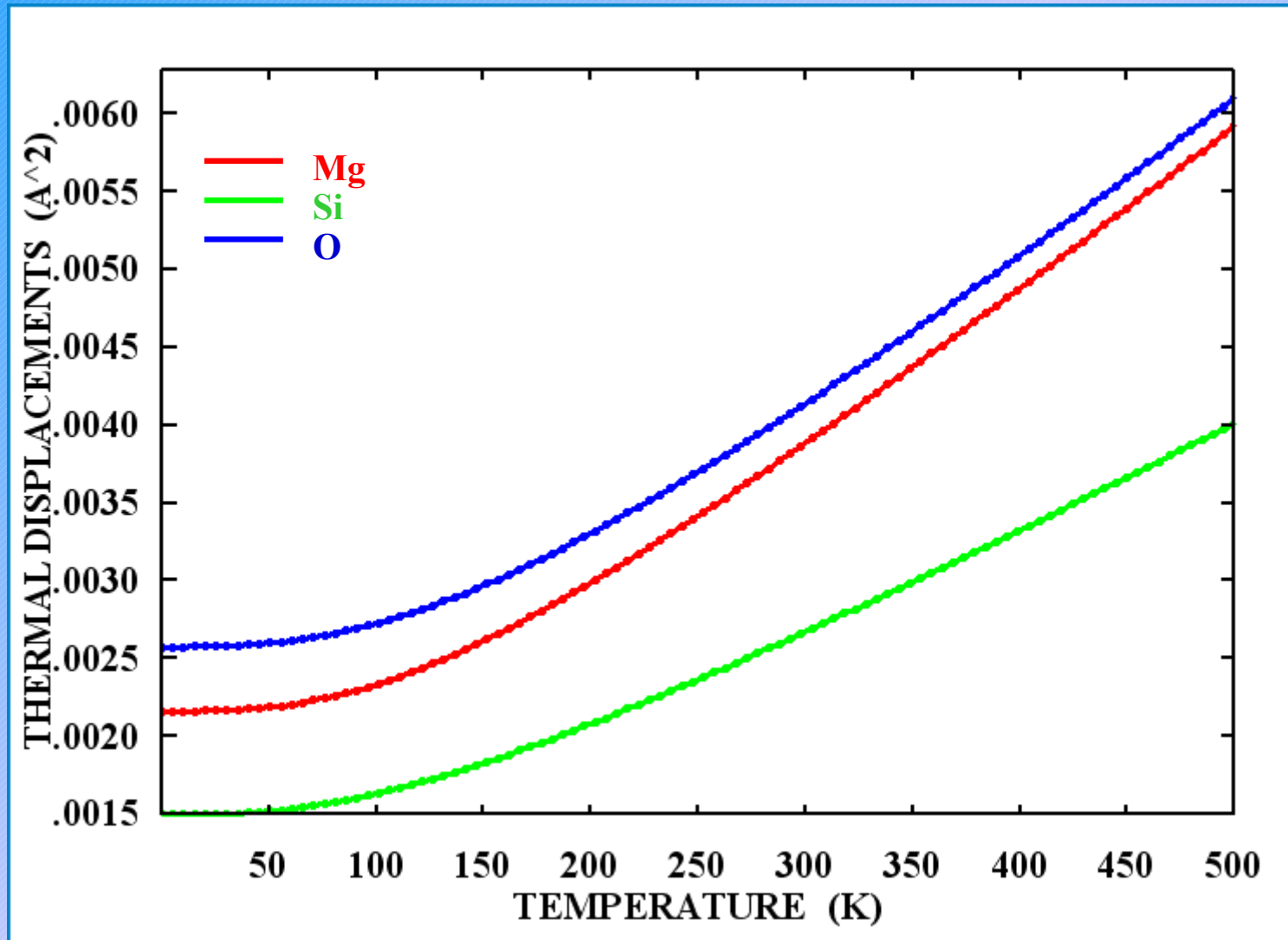


Thermodynamical Functions from harmonic phonons





Mean square displacements $\langle U^2 \rangle$



Create/Dispersion Curves/Plot for Atoms - filtering of phonon branches

$\mathbf{e}_i(\mathbf{k}, j; \mu)$ - i-components (x,y,z) of polarization vector of atom μ ,
corresponding to phonon wave vector $\mathbf{k}$ from branch j .



$$\sum_{\mu,i} |\mathbf{e}_i(\mathbf{k}, j; \mu)|^2$$

Independent contribution to phonon branch of sum of **selected** degree of freedom. Intensity **does not depend** on B.Z.

$$\left| \sum_{\mu,i} \frac{\mathbf{e}_i(\mathbf{k}, j; \mu)}{\sqrt{M_\mu}} \right|^2$$

Coherent contribution of degree of freedom. Indicates real occupation of phonon branches. Intensity is independent of selection of unit cell (smaller, larger). Intensity **depends** on B.Z.

$$\frac{1}{k^2} \left| \sum_{\mu,i} \frac{\mathbf{k} \cdot \mathbf{e}_i(\mathbf{k}, j; \mu)}{\sqrt{M_\mu}} \right|^2$$

Coherent contribution of degree of freedom projected on the transmission wave vectors $\mathbf{k}$ as in coherent inelastic neutron scattering. Intensity **depends** on B.Z.

Filters permit to establish an interpretation of a given phonon mode !!!

$$\omega^2(\mathbf{k}) \mathbf{e}(\mathbf{k}) = \mathbf{D}(\mathbf{k}) \mathbf{e}(\mathbf{k})$$

$$\sum_{\mu,j} |\mathbf{e}_i(\mathbf{k}, j; \mu)|^2 = 1$$

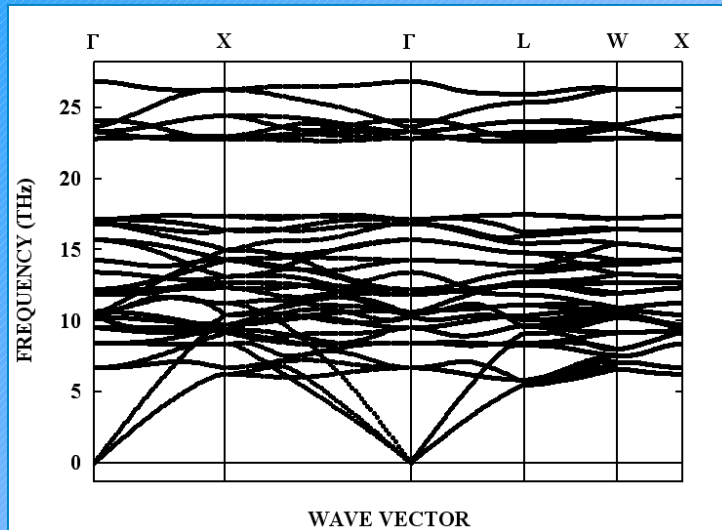
$\mu, j = \text{all}$

Normalization condition

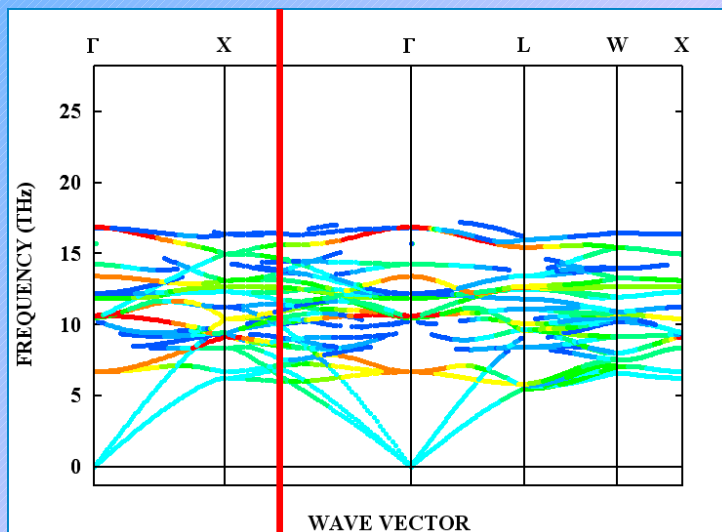
Mg₂SiO₄ Participation of atom in phonon dispersion relations:

$$\sum_{\mu,i} |e_i(\mathbf{k}, j; \mu)|^2$$

$$\omega^2(\mathbf{k}) \mathbf{e}(\mathbf{k}) = \mathbf{D}(\mathbf{k}) \mathbf{e}(\mathbf{k})$$

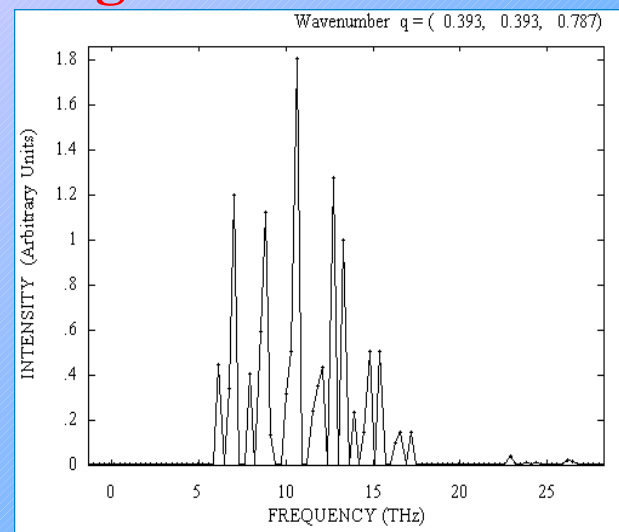


Total - no filter

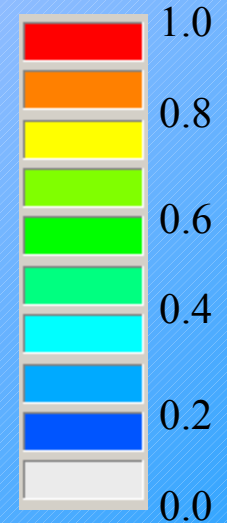


Mg

Intensity



Intensity scale (%)

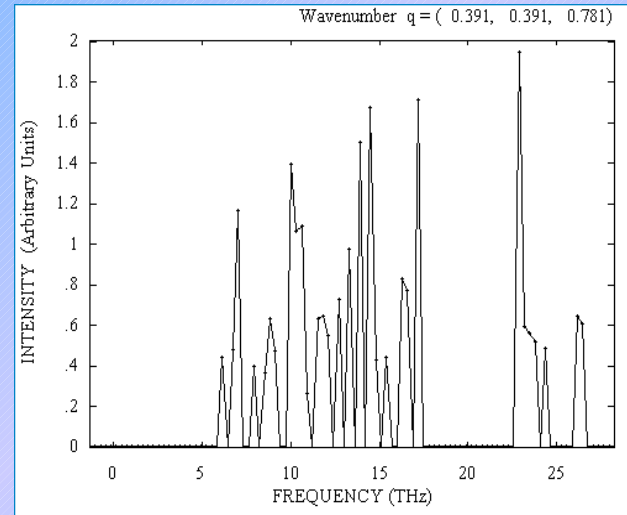
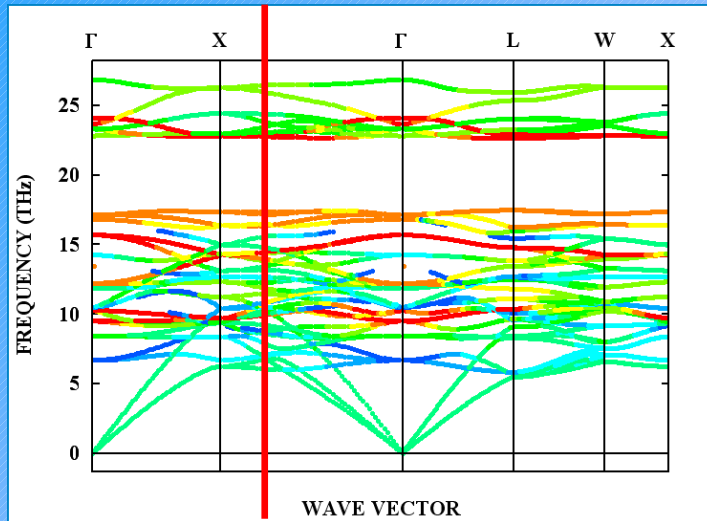


Mg₂SiO₄ Participation of atom in phonon dispersion relations:

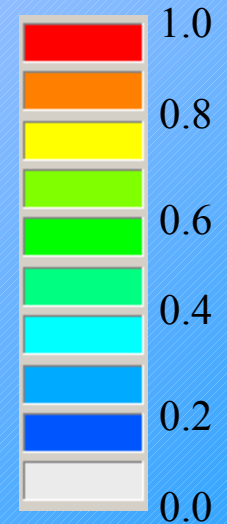
$$\sum_{\mu,i} |\mathbf{e}_i(\mathbf{k}, j; \mu)|^2$$

O

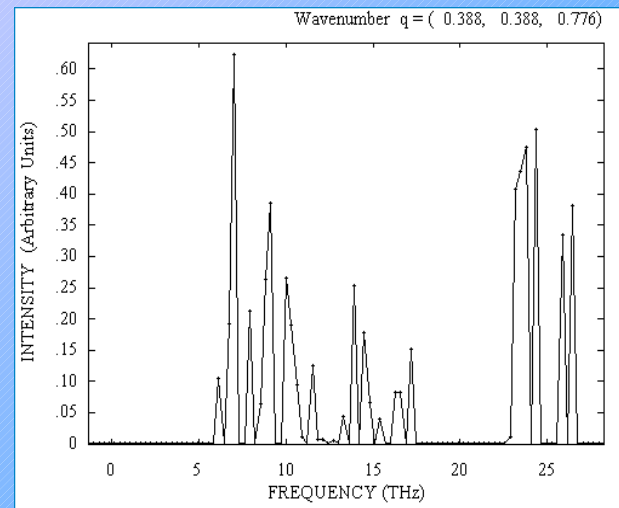
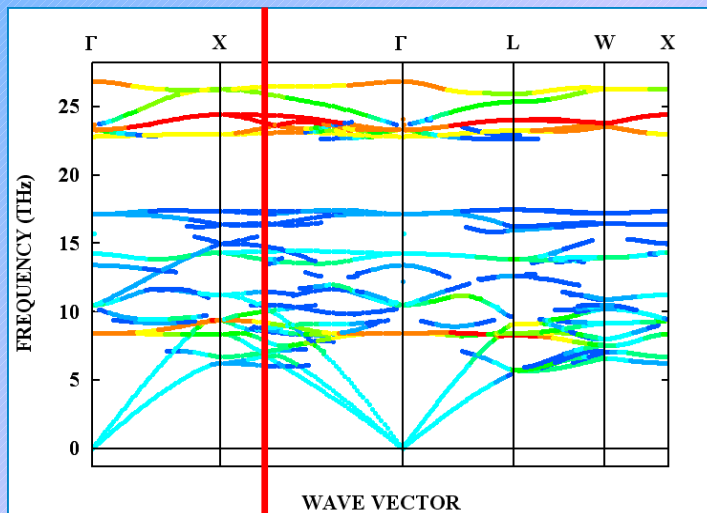
Intensity



Intensity scale (%)



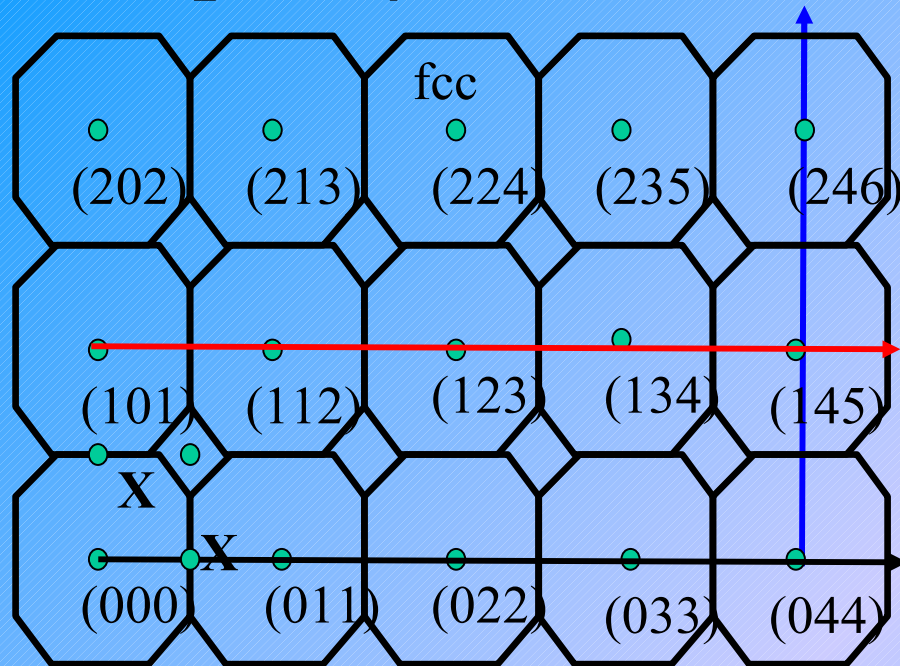
Si



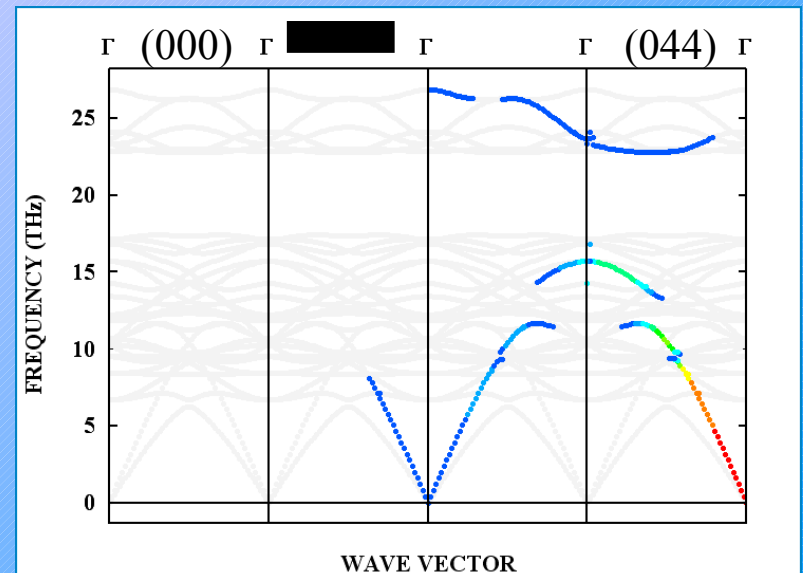
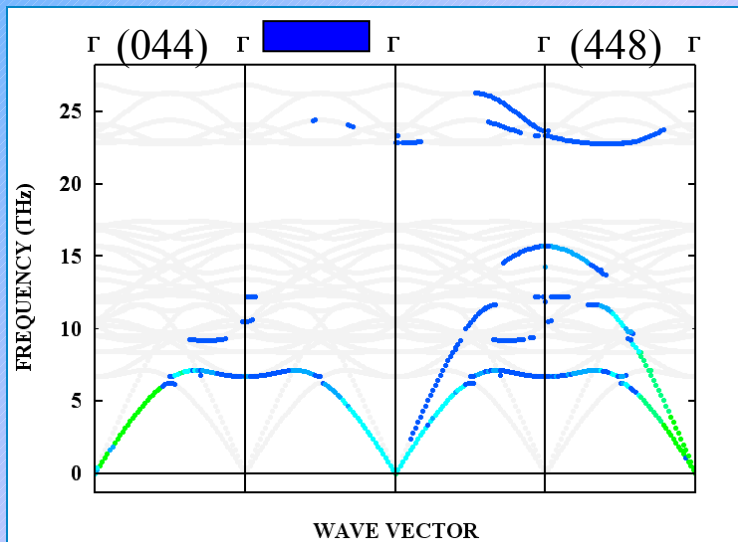
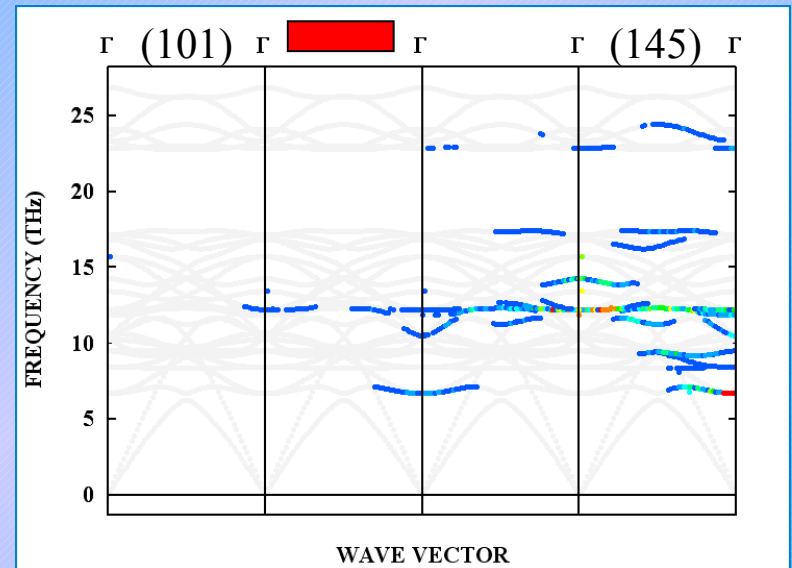
**Phonon intensity in inelastic
coherent neutron scattering:**



Coherent inelastic
neutron scattering:



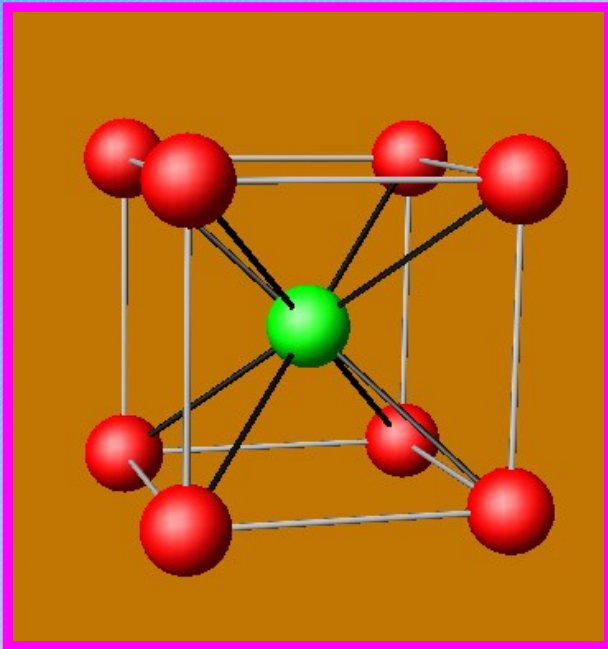
$$\mathcal{F}(\mathbf{k}, j) = \left| \sum_{\mu} a_{coh}(\mu) \exp[-W_{\mu}(\mathbf{k})] \frac{2\pi \mathbf{k} \cdot \mathbf{e}(\mathbf{k}, j; \mu)}{\sqrt{M_{\mu}}} \right|^2$$



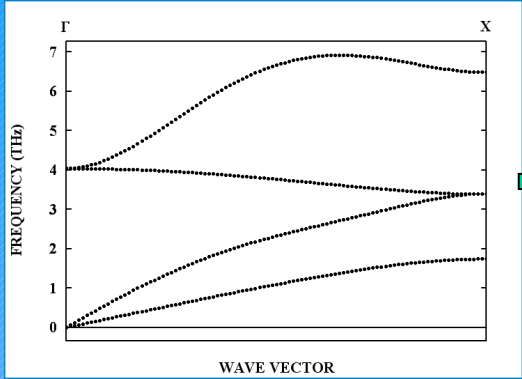
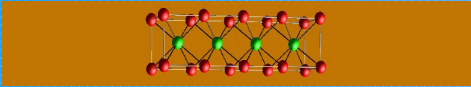


Elongated supercell.

**LO / TO splitting. Part 1.
Result of LO mode coupling
to induced macroscopic
electric field.
Model of simple cubic CsCl**

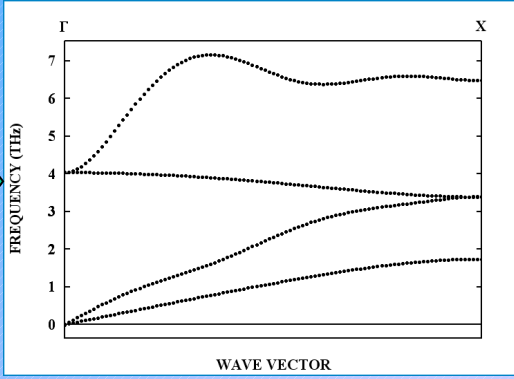
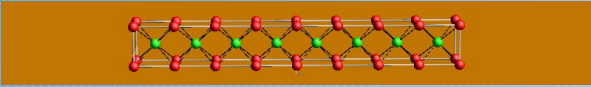


*** CsCl**
1x1x4



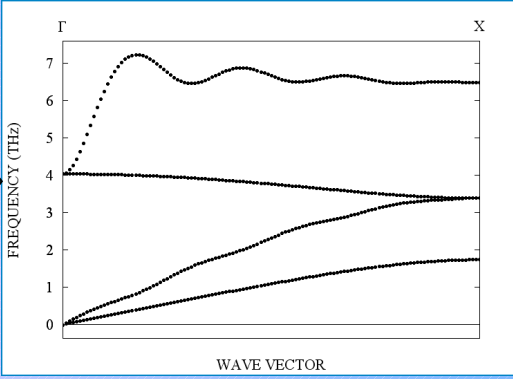
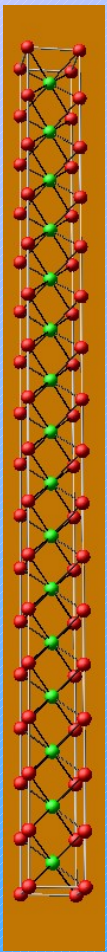
NO non-analytical term

1x1x8



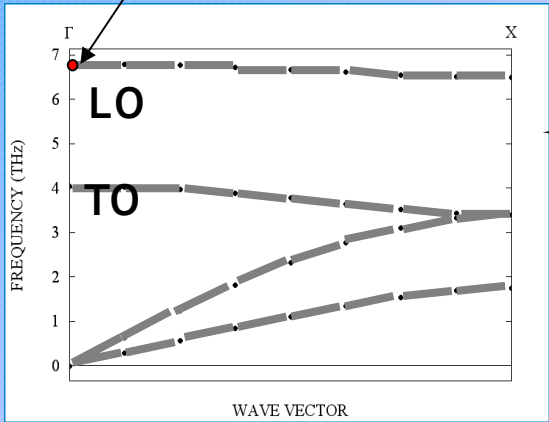
NO non-analytical term

1x1x16



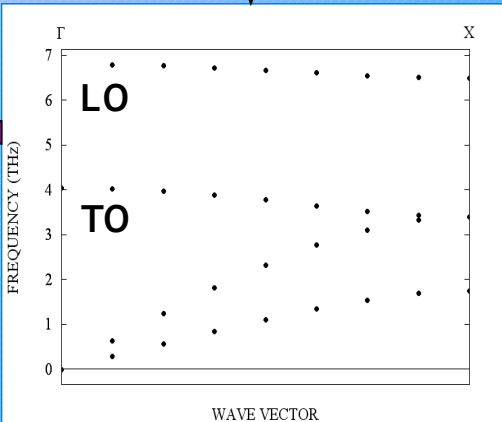
NO non-analytical term

LO from non-analytical term



WITH non-analytical term

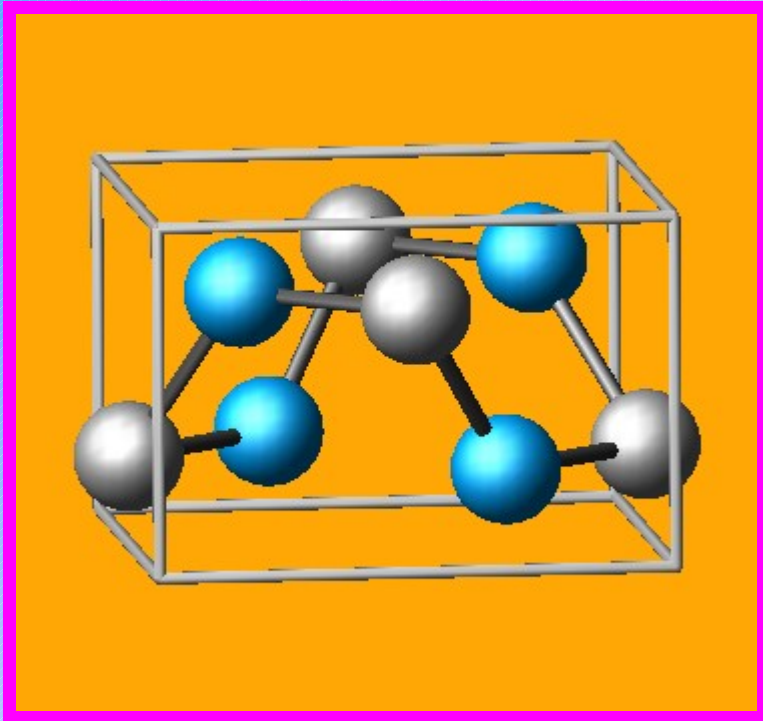
**Add non-analytical term
to dynamical matrix**



NO non-analytical term



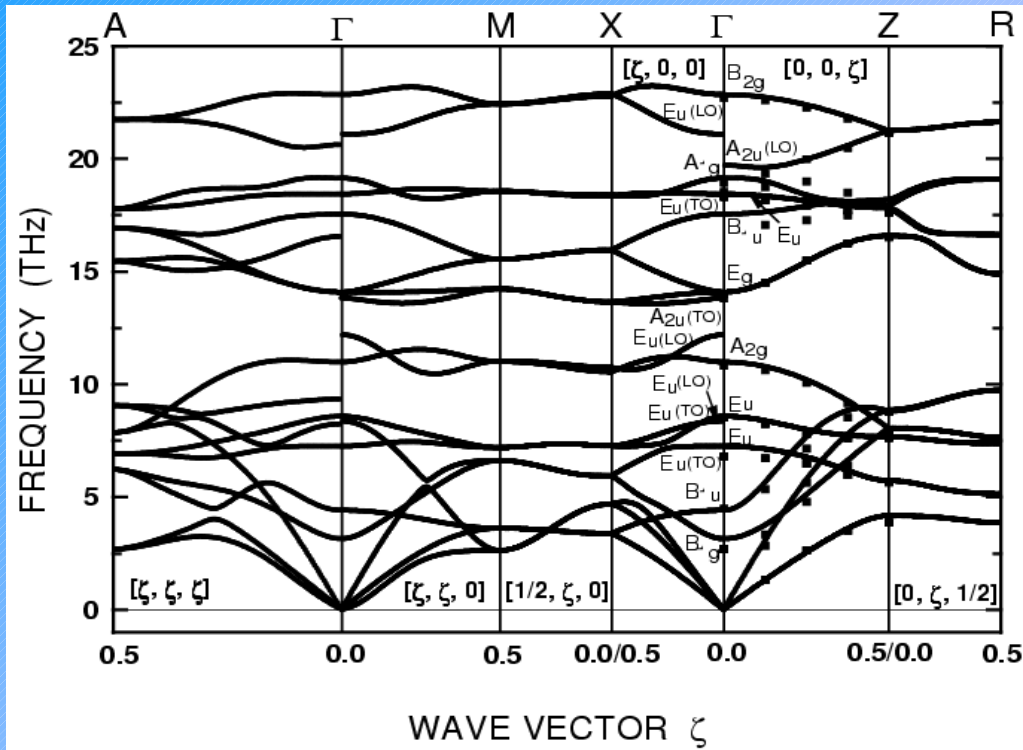
LO /TO splitting. Part 2. Tetragonal rutile crystal SnO_2



LO/TO splitting from singular term

SnO₂

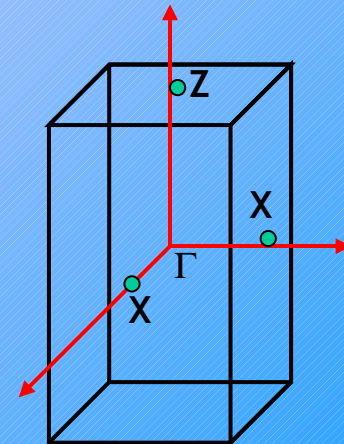
Rutile (P4₂/mm)



Add non-analytical term to classical dynamical matrix

$$D_{\alpha\beta}^M(\mathbf{k}; \mu\nu) = D_{\alpha\beta}(\mathbf{k}; \mu\nu) + \frac{4\pi e^2}{V\epsilon_{\infty}\sqrt{M_{\mu}M_{\nu}}} \frac{[\mathbf{k} \cdot \mathbf{Z}^*(\mu)]_{\alpha}[\mathbf{k} \cdot \mathbf{Z}^*(\nu)]_{\beta}}{|\mathbf{k}|^2} \times \exp[-2\pi i \mathbf{g} \cdot (\mathbf{r}(\mu) - \mathbf{r}(\nu))]$$

$$\times d(\mathbf{q}) \exp \left\{ -\pi^2 \left[\left(\frac{k_x}{\rho_x} \right)^2 + \left(\frac{k_y}{\rho_y} \right)^2 + \left(\frac{k_z}{\rho_z} \right)^2 \right] \right\}$$



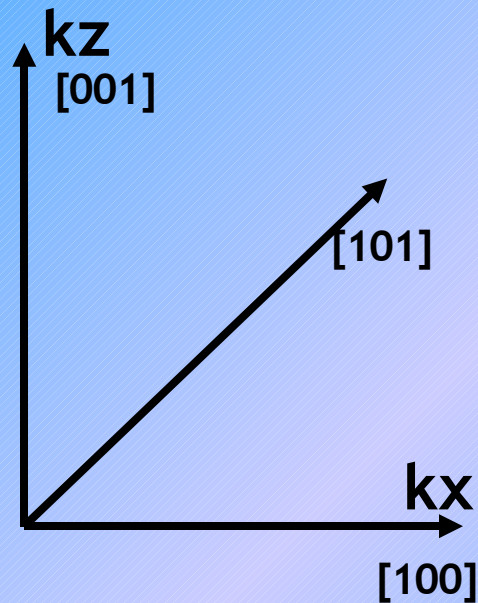
Supercell: 2x2x2, 48 atoms.
Points at $\Gamma - Z$ are from elongated
supercell 1x1x8

K.Parlinski and Y.Kawazoe, Euro.Phys.J. **B13**, 679 (2000)

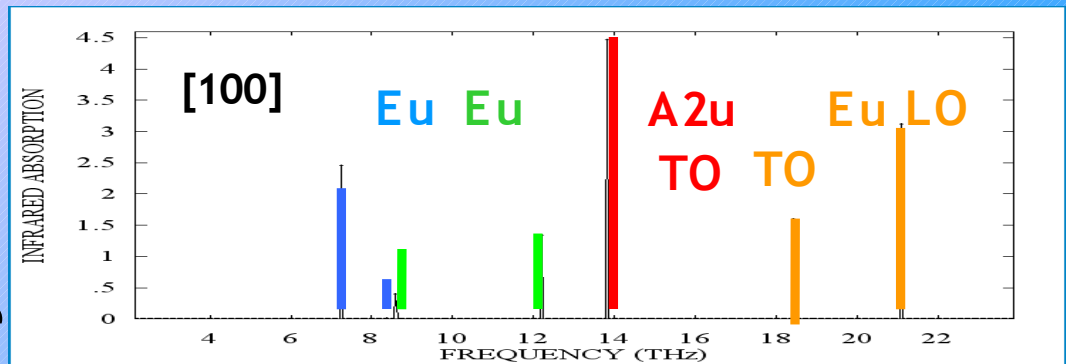
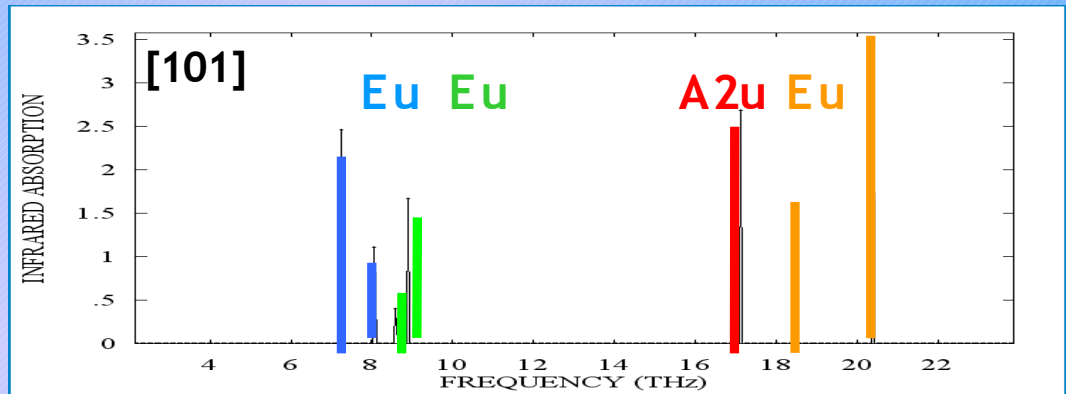
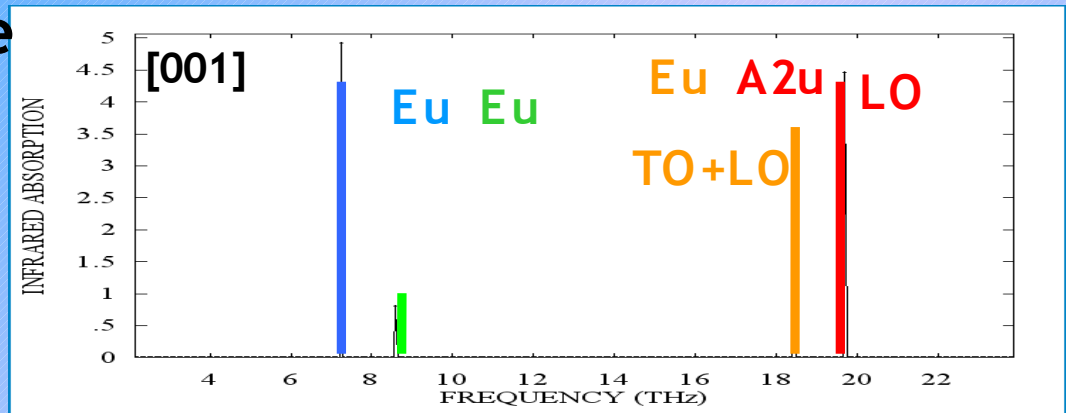
Example: SnO₂ S.C.: 1x1x8

SnO_2 Right movie

Infrared absorption
depends on
wavevector orientation

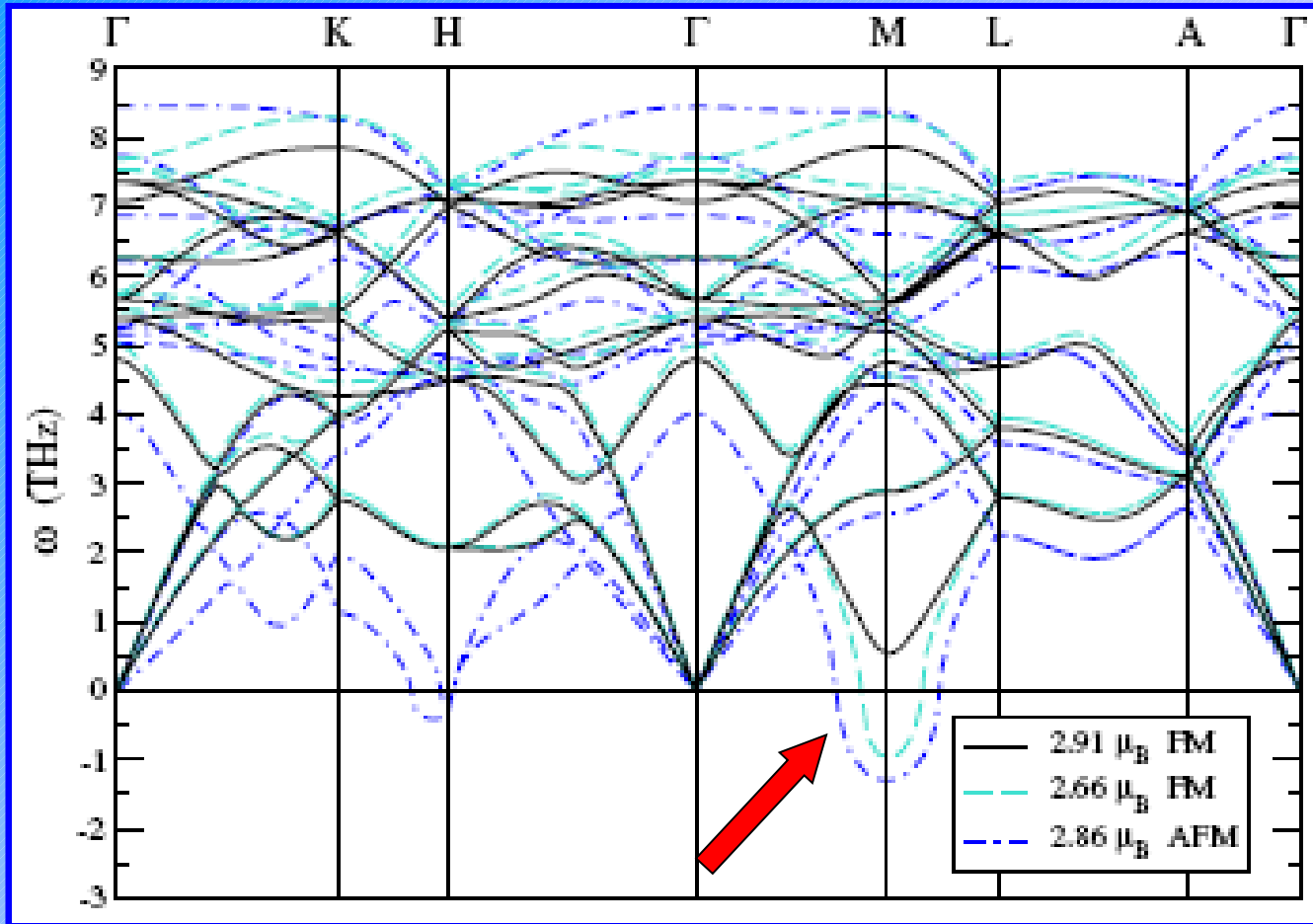


Left movie



Magnetism and Phonons in MnAs

MnAs



Soft mode at M point depends on

- magnitude of magnetic moment
- FM, or AFM ordering

Calculations performed with **fixed value of magnetic moment μ_B**

Hexagonal structure, P63/mmc, room temperature.
Supercell 2x2x1 with 16 atoms, VASP.

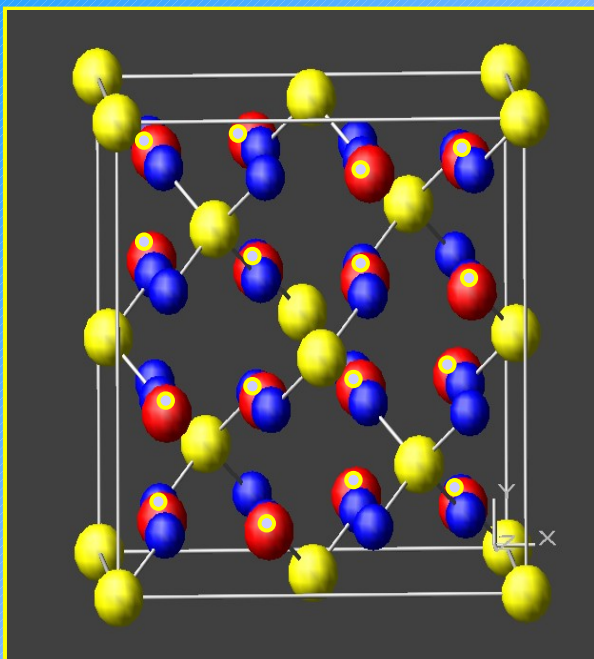
Some reasons why phonon dispersion curves are wrong, i.e. have imaginary (negative) frequencies

- Magnetic system is treated without magnetic interaction, or with wrong magnetic structure.
- Interatomic potential range must be taken longer.
- Exists a proper soft mode, and imaginary frequency is correct.
- Electron correlation is neglected.
- ?



FM

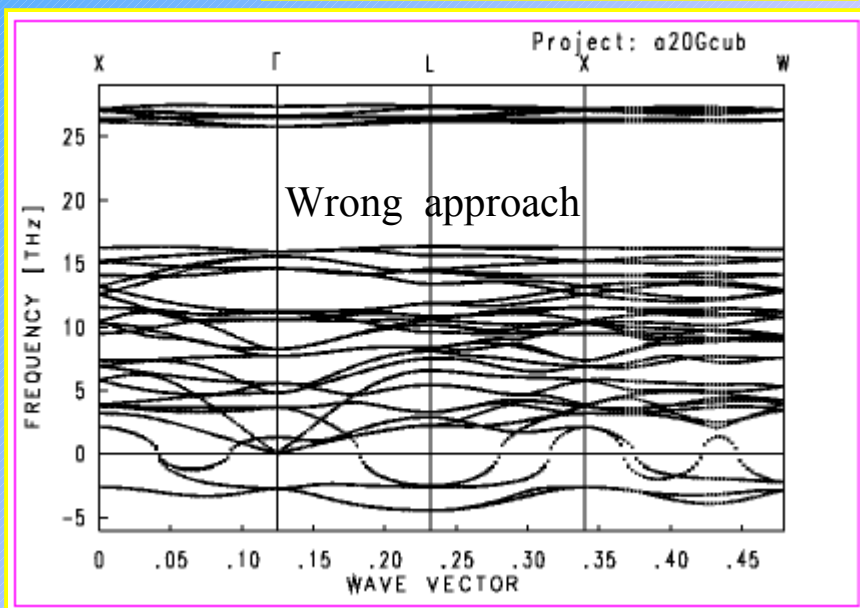
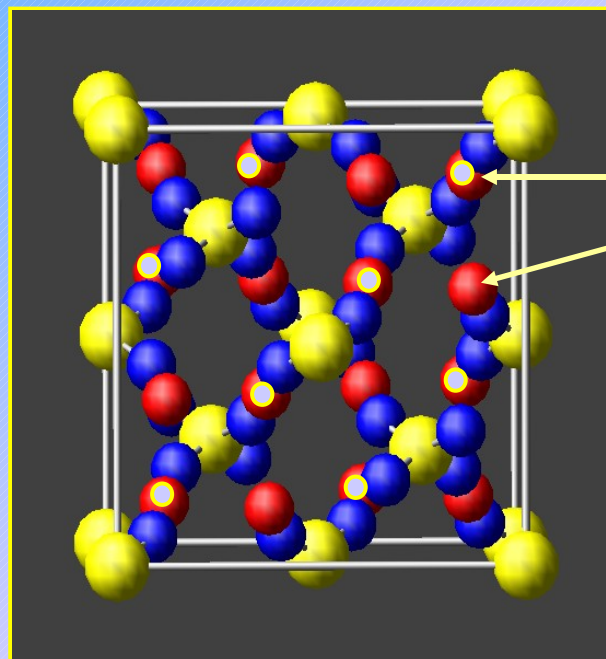
3 HF force
lists as
required by
cubic
Fd-3m



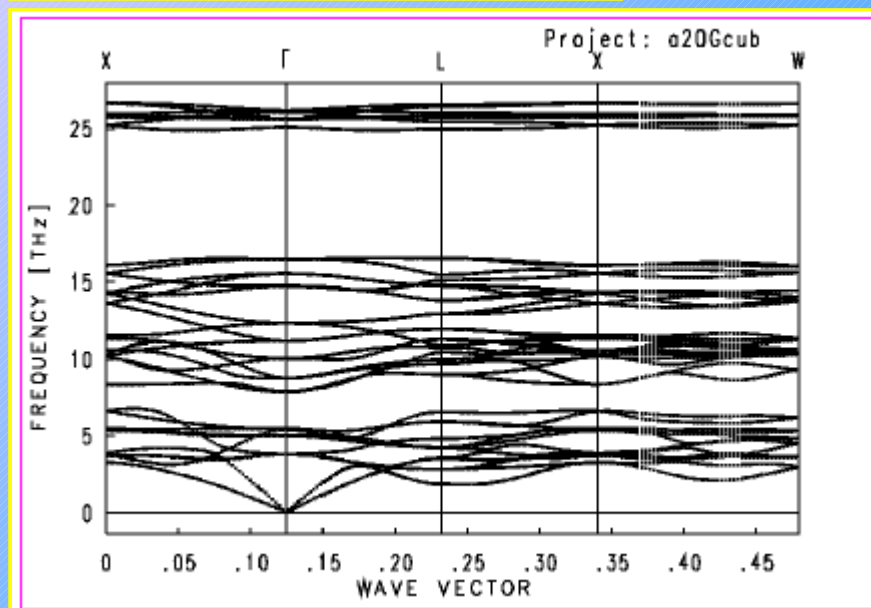
AF

up spin
down spin

8 HF force
lists as
required by
tetragonal
I4₁/amd



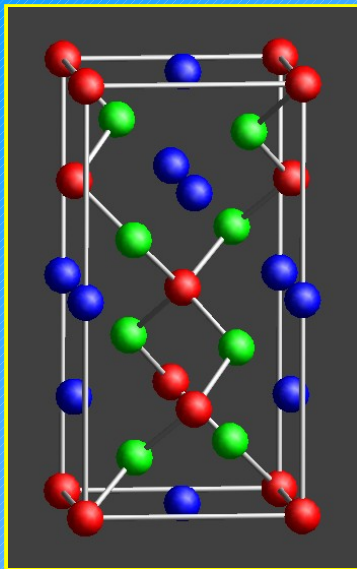
cubic



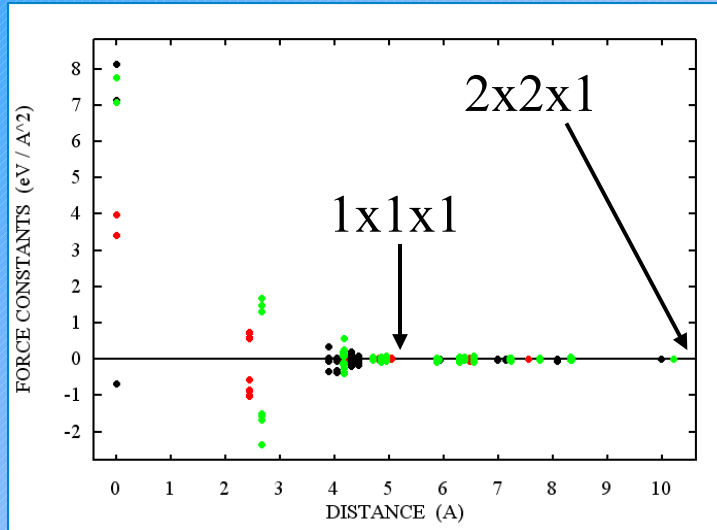
slightly tetragonal

Short interatomic potential

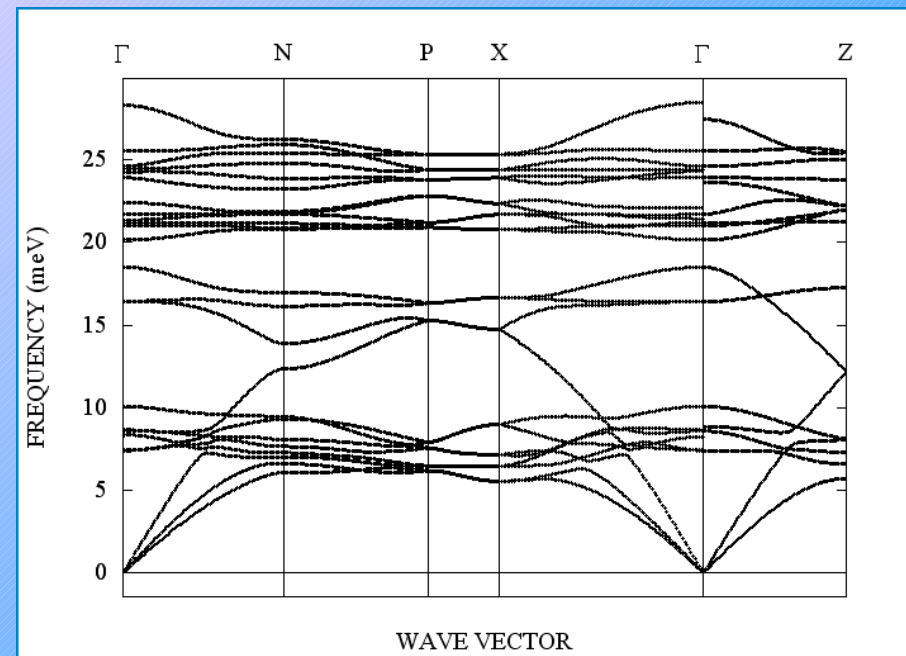
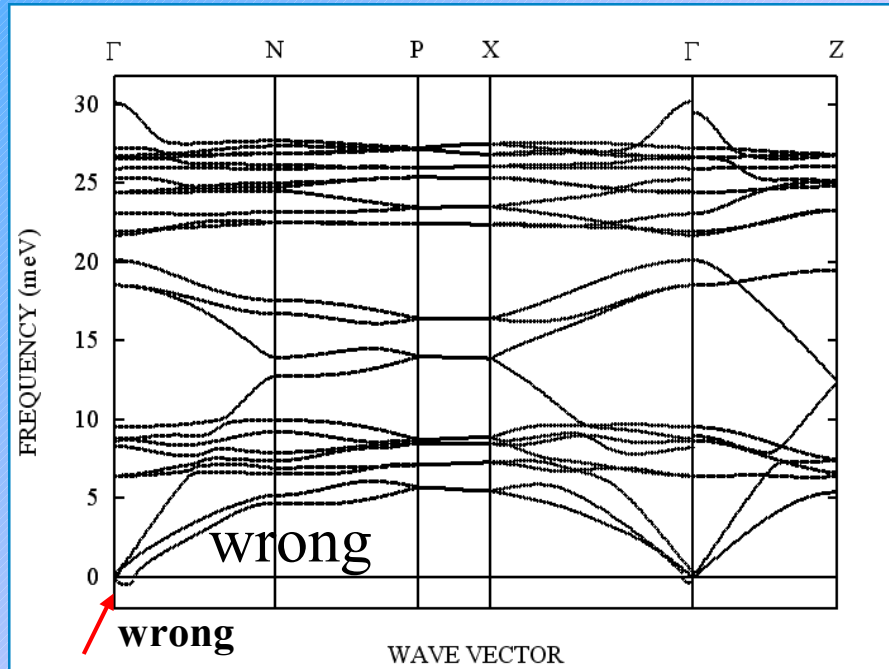
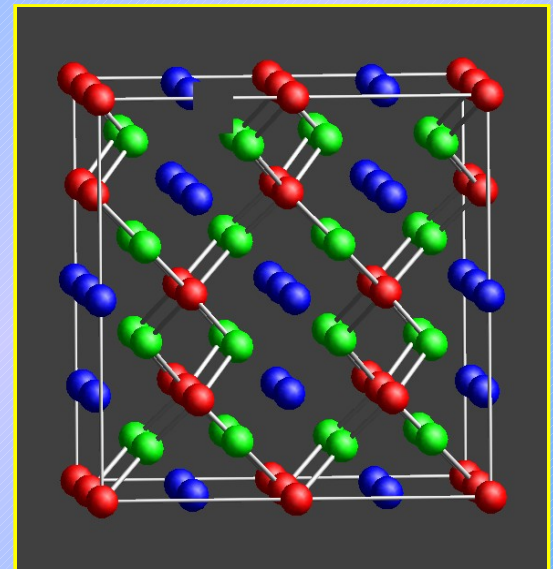
1 x 1 x 1



CuInS_2



2 x 2 x 1



J.Lazewski, H.Neumann, K.Parlinski, G. Lippold, and B.J. Stanbery, Phys.Rev. B 68, 144108 (2003).

Space subgroups induced by a phonon mode

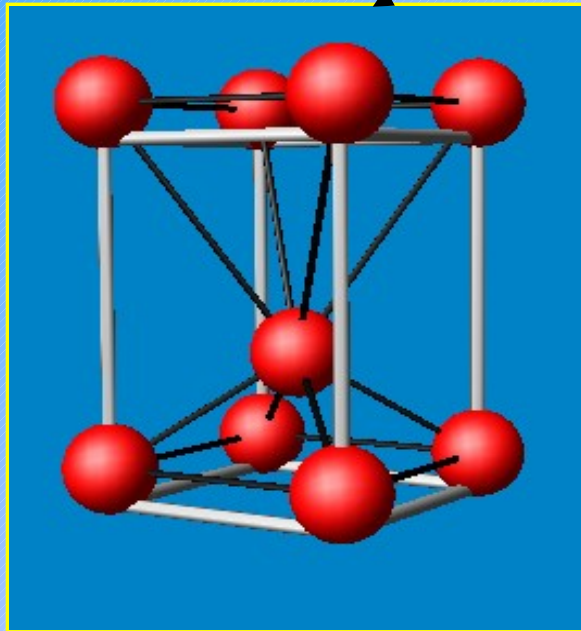
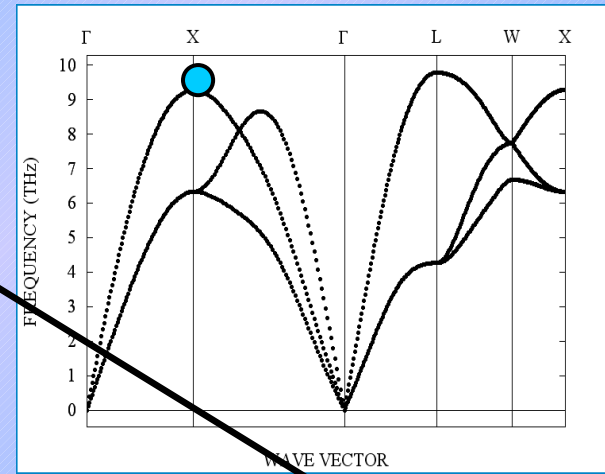
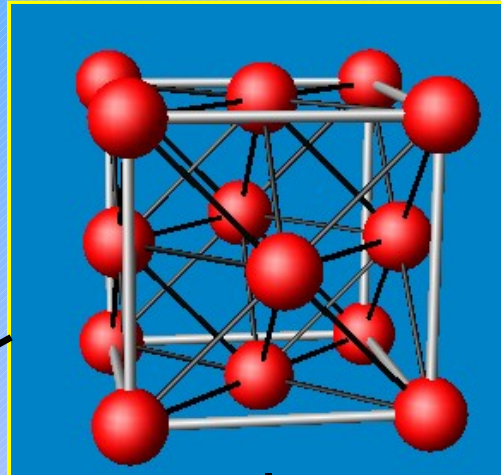
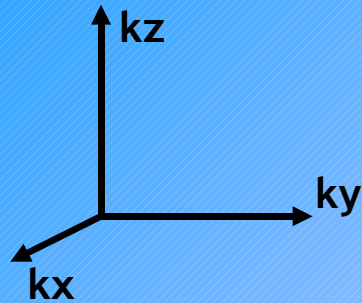
Space group analysis can be useful in:

- structural phase transitions**
- linear electron-phonon coupling**
- magnetic-phonon coupling**

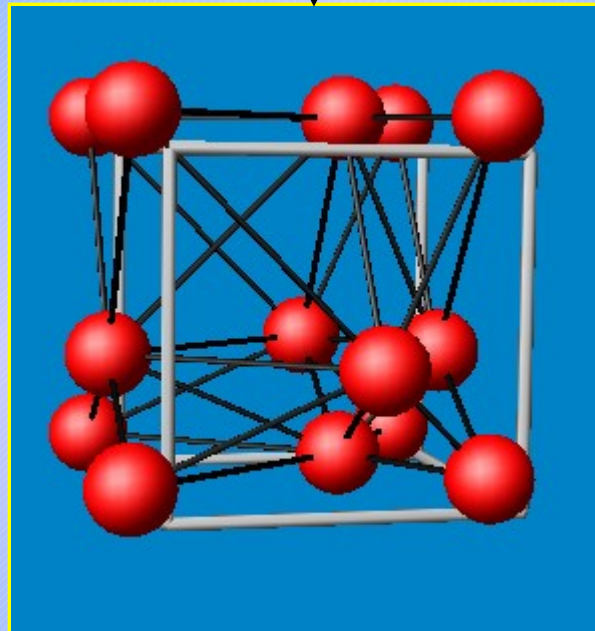
Example: Space subgroups induced by a phonon mode „X3-” in fcc Al

Fm-3m, No=225, (000)

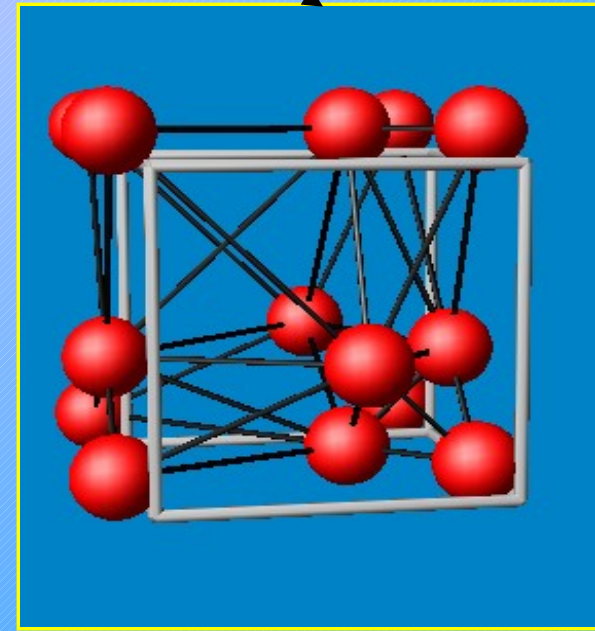
Irreducible star {kx} has 3 arms



P4/nmm, No=129, (a00)



P4₂/mcm, No=132, (aa0)



P-43m, No=215, (aaa)

Towards crystals with point defects

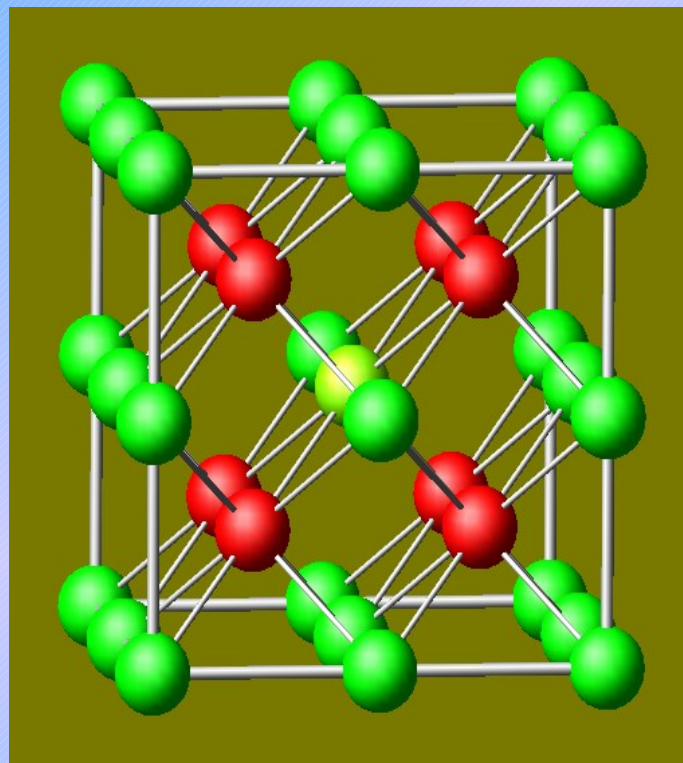
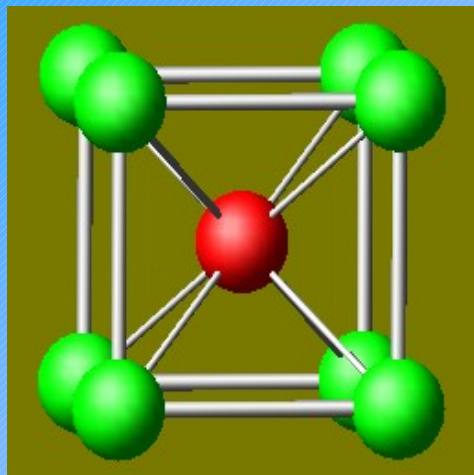
Is description of phonon dispersion curves of primitive and supercell equivalent?

Yes, it is, but filters should be used.
Example.

Model crystal AB

2x2x2

1x1x1



Reminder:

$$\frac{1}{k^2} \left| \sum_{\mu,i} \frac{\mathbf{k} \cdot \mathbf{e}_i(\mathbf{k}, j; \mu)}{\sqrt{M_\mu}} \right|^2$$

Average over
orientations



$$\left| \sum_{\mu,i} \frac{\mathbf{e}_i(\mathbf{k}, j; \mu)}{\sqrt{M_\mu}} \right|^2$$

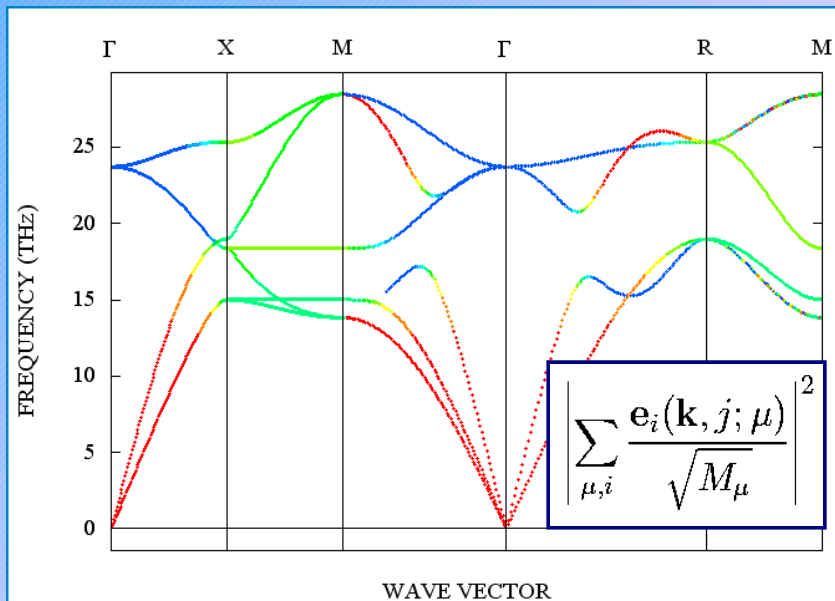
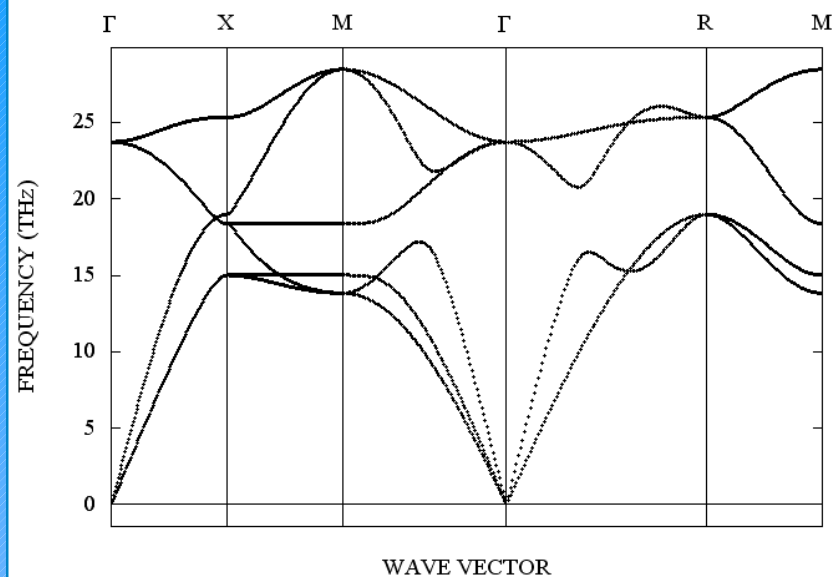
**Intensity of coherent neutron scattering averaged
over orientations (k.e) gives Averaged Filter (AF)**

AF

Model crystal AB

1x1x1

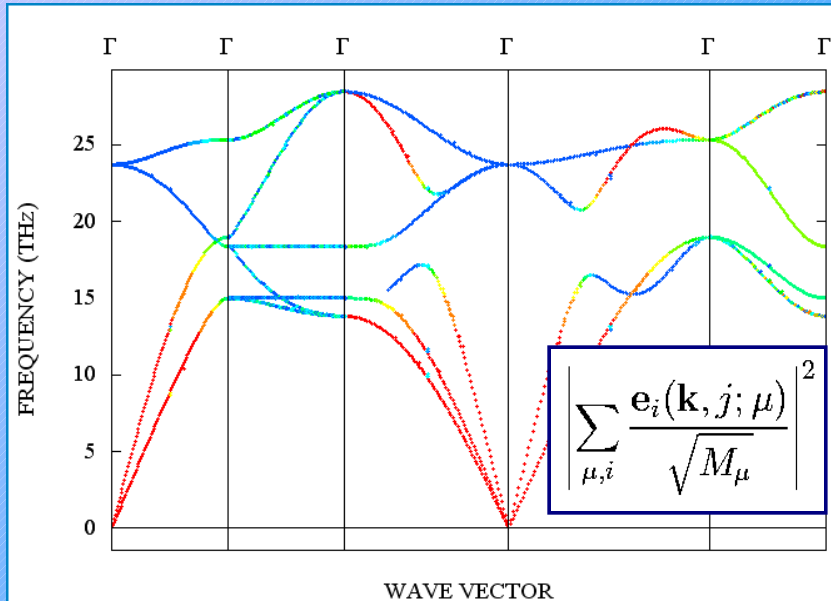
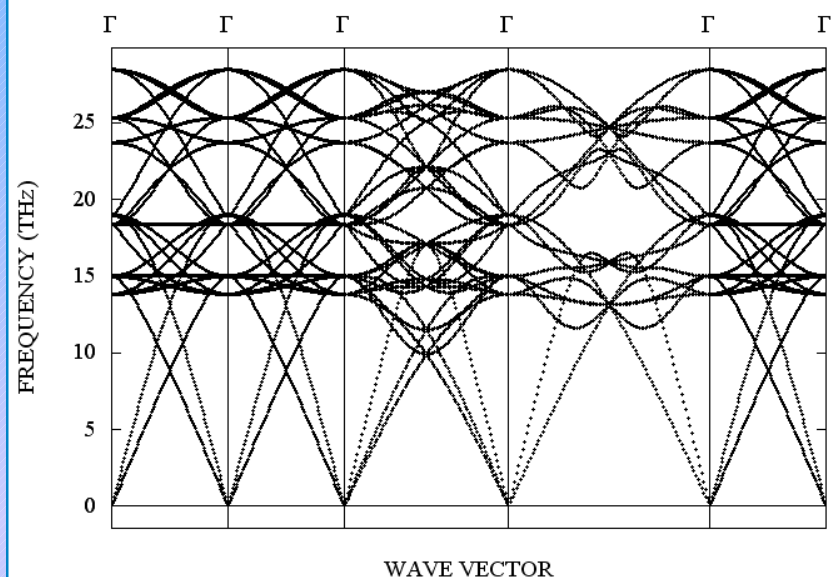
Phonon branches: 6



2x2x2

Phonon branches: 48

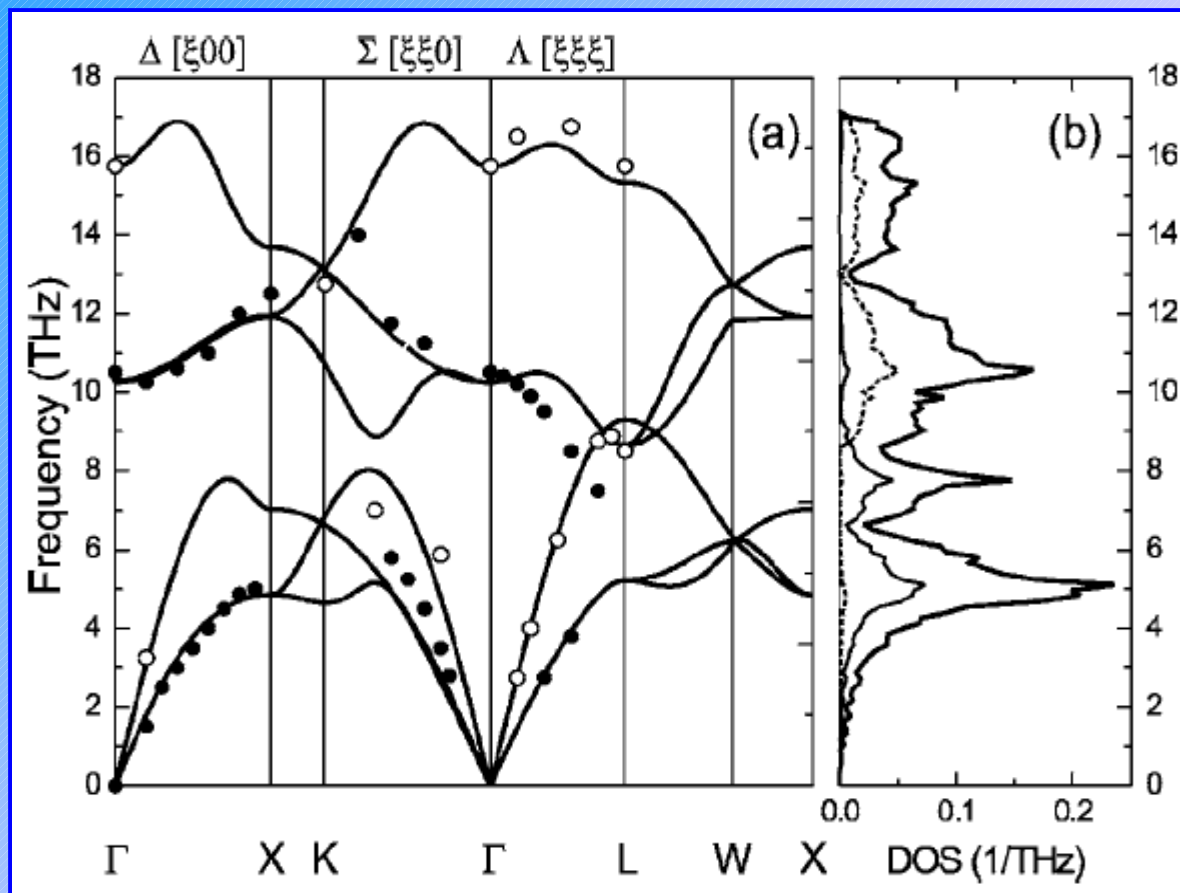
Many phonon modes are inactive



CoO crystal

Strong electron correlation. LDA+U, GGA+U

PAW, GGA+U, Local Coulomb repulsion $U = 7.1$ eV
Hund's exchange $J = 1.0$ eV

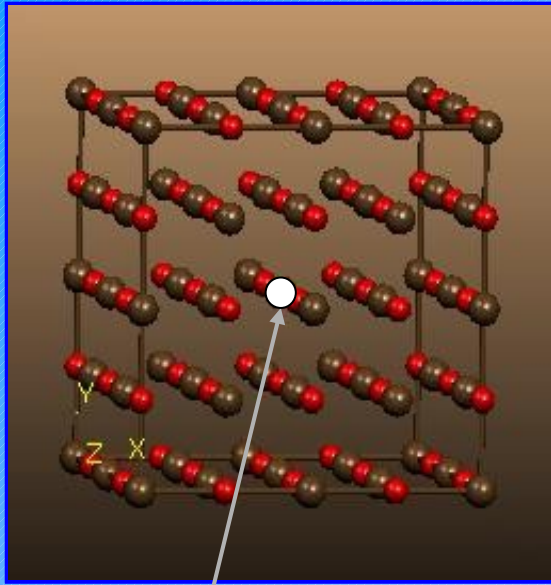


**cubic NaCl -
type**
Phonon dispersion curves
calculated for $U=0$ and $J=0$
shows imaginary frequencies

Experiment: J.Sakurai, W.J.L.Buters, R.A.Cowley, and G.Dolling, Phys.Rev. **167**, 510 (1968)

Calculations: U.D.Wdowik, and K.Parlinski, PRB, 75, 104306 (2007)

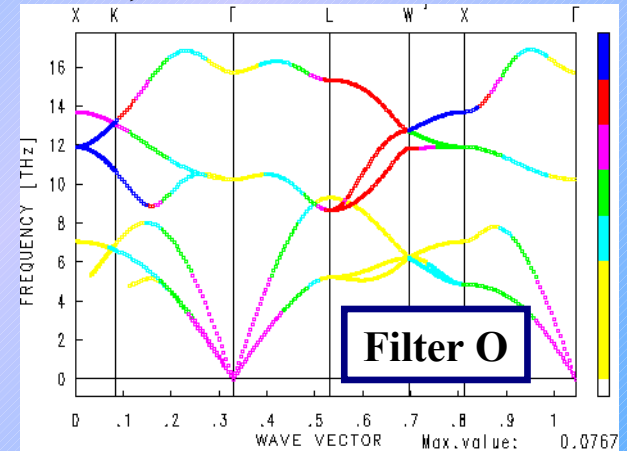
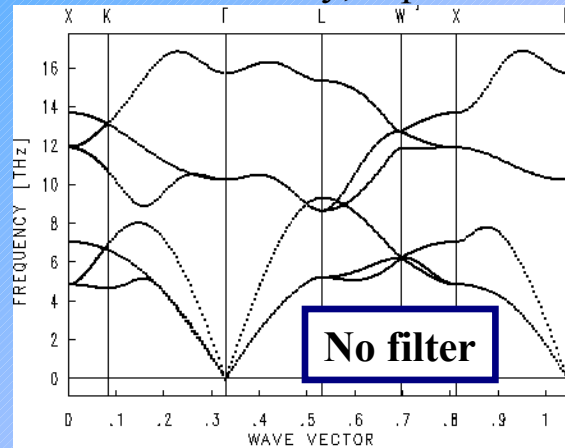
CoO, cubic type of NaCl Crystal CoO with Co vacancy



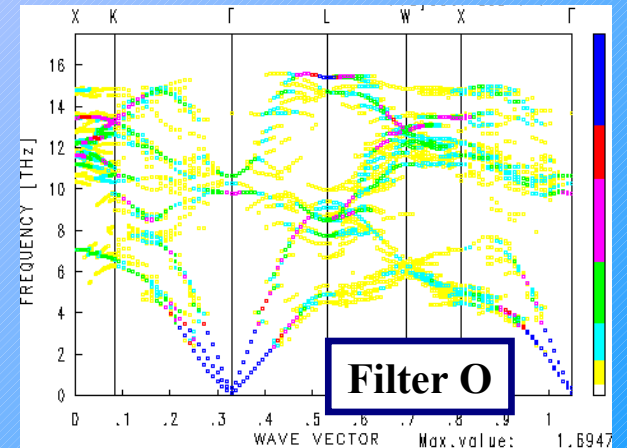
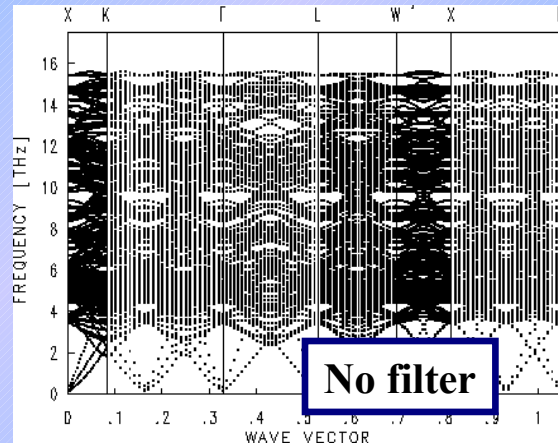
Cobalt vacancy of concentration ~3%

VASP, LDA+U
Local Coulomb repul. $U = 7.1$ eV
Hund's exchange $J = 1.0$ eV

CoO no vacancy, supercell of 64 atoms, 6 modes



CoO with vacancy at Co, supercell of 63 atoms, 189 modes



Filter O

Intensity
colour



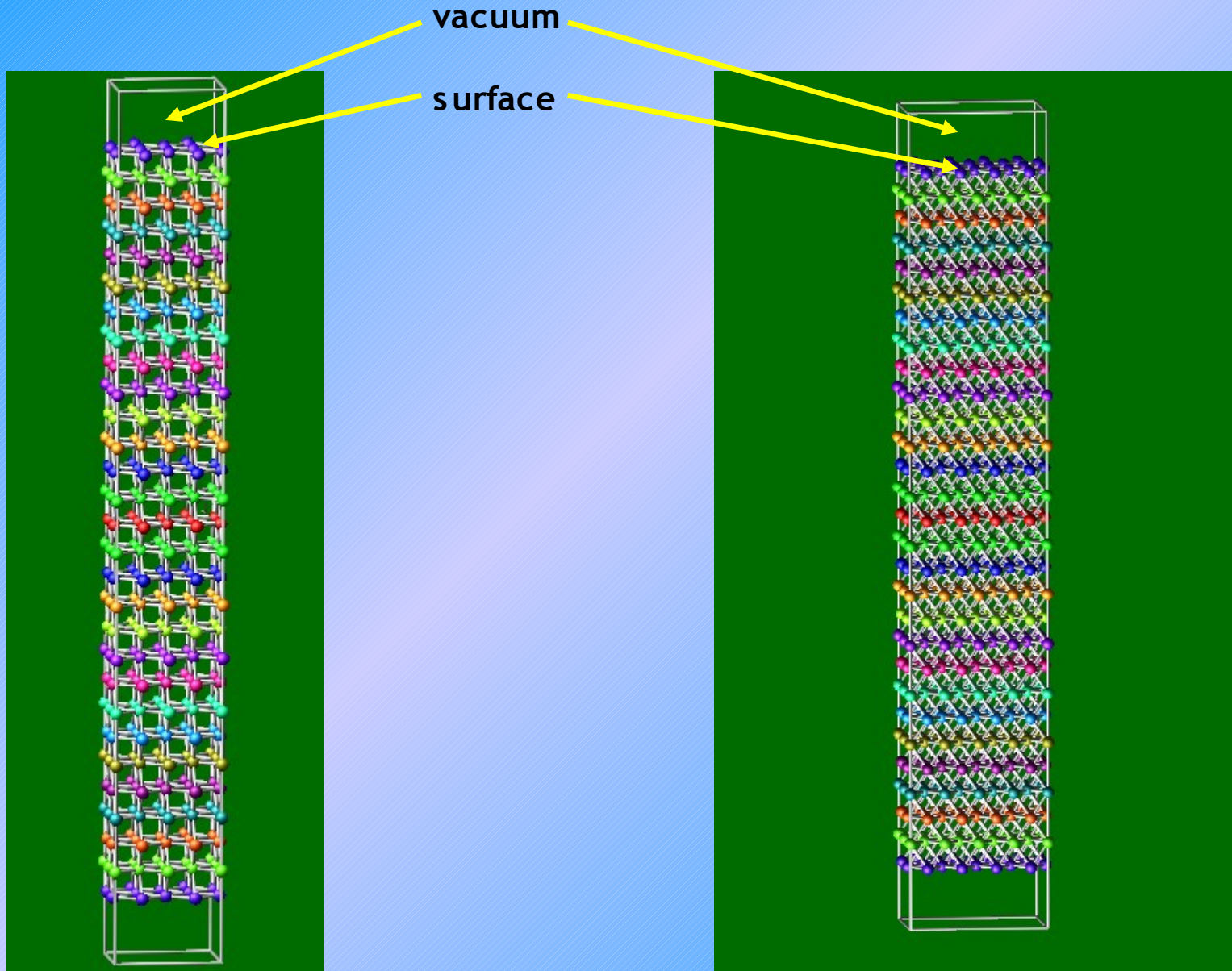
$$\left| \sum_{\mu, i} \frac{\mathbf{e}_i(\mathbf{k}, j; \mu)}{\sqrt{M_\mu}} \right|^2$$

Indicates occupation of phonon branches. Independent on size of unit cell. Depends on B.Z.

Phonons on surface

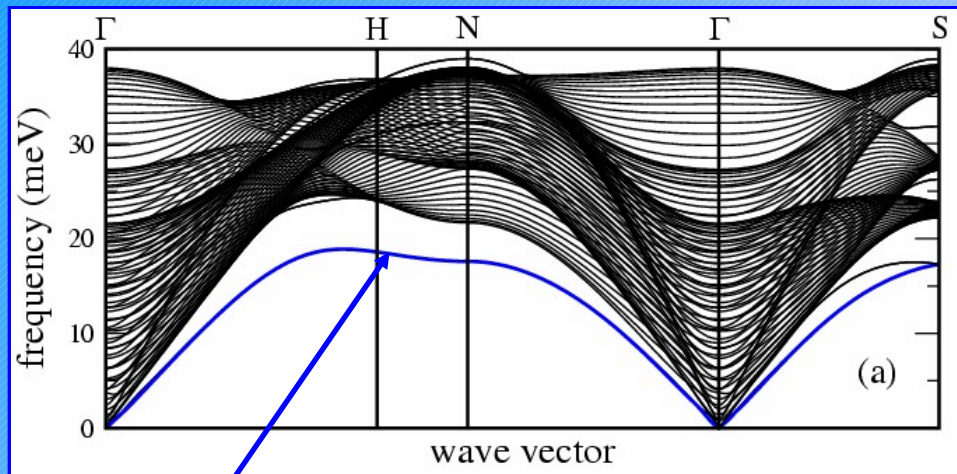
Fe (110)

Supercell with 3d periodic boundary conditions for surface calculations



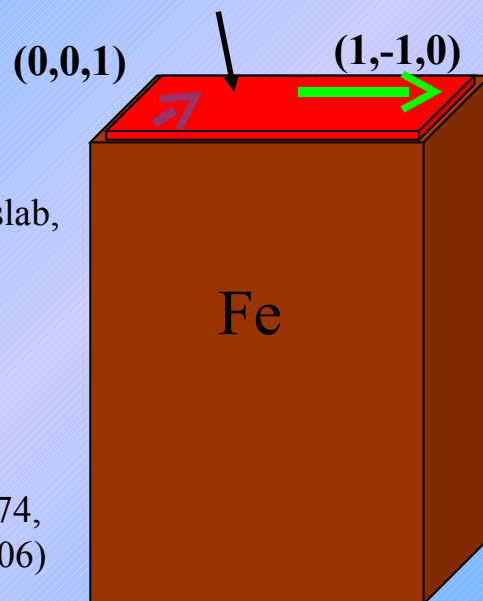
Phonons on Fe (110) surface

Slab of 29 layers of Fe (110)



Rayleigh mode

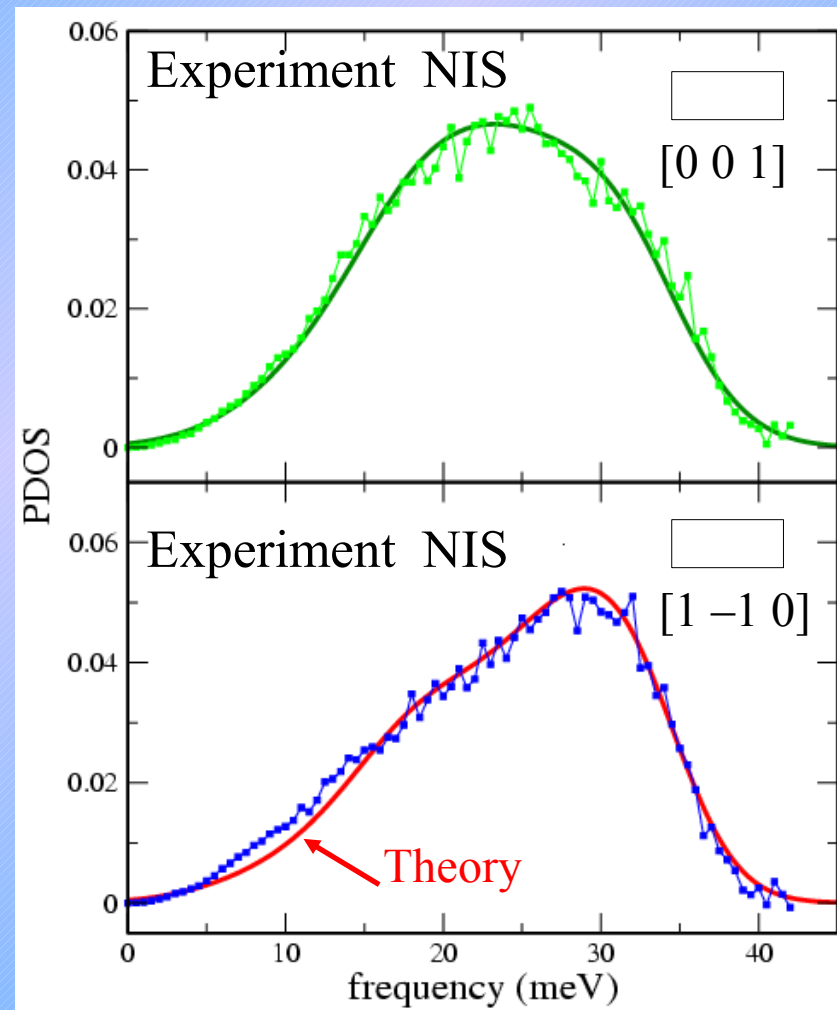
Fe^{57} single crystal



Filling slab approach:

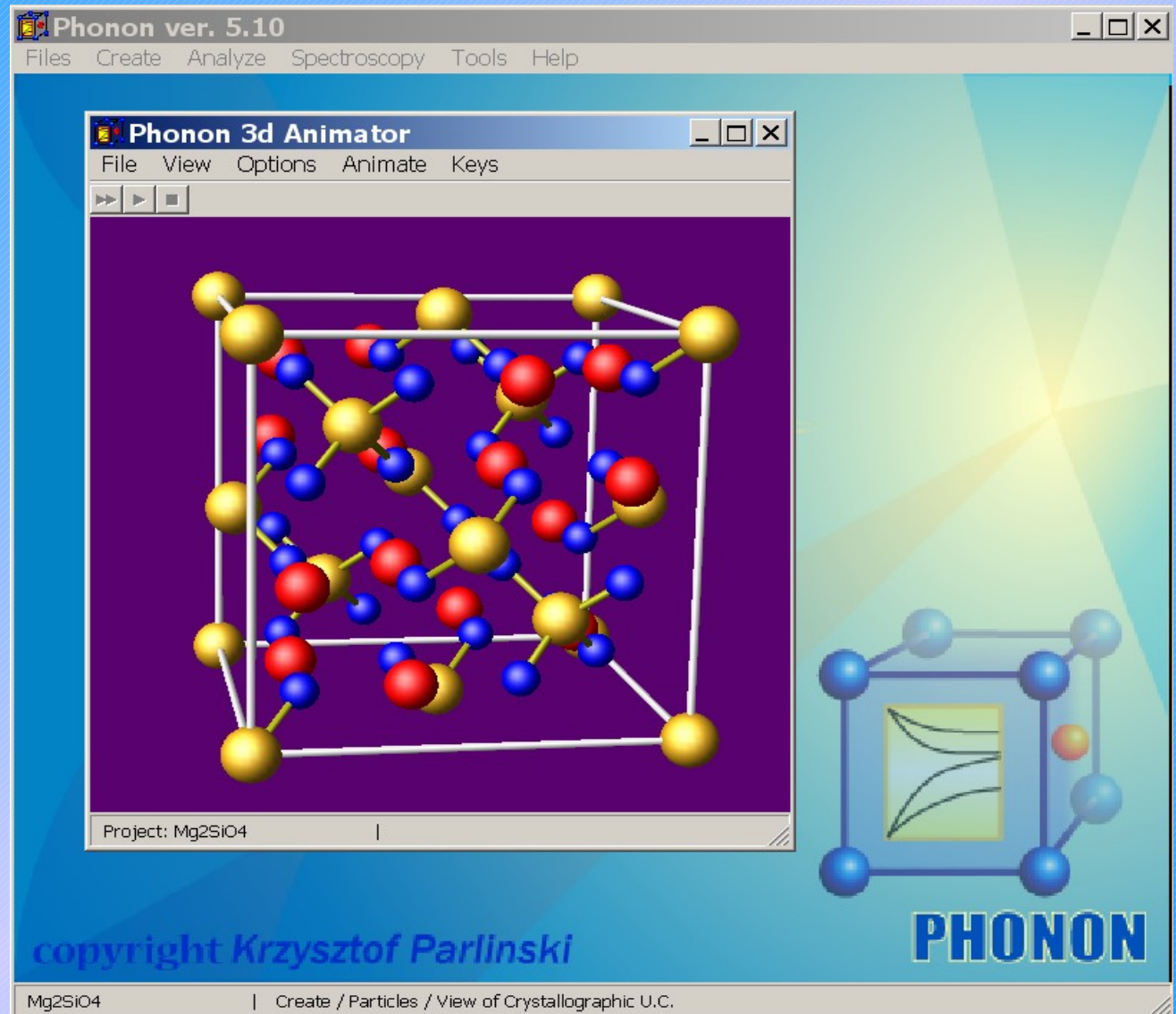
5 layers (110) of surface slab,
6 times 4 slabs of bulk Fe

K. Parlinski, Phys. Rev. B 74,
184309 (2006)



T. Ślęzak, J. Łażewski, S. Stankov, K. Parlinski, R. Reitering, M. Rennhofer, R. Ruffer, B. Sepiol, M. Ślęzak, N. Spiridis, M. Zając, A. I. Chumakov, and J. Korecki, Phys. Rev. Lett. 99, 066103 (2007)

Phonon Software, ver. 5.10



<http://www.ComputingForMaterials.com/>

Copyright Krzysztof Parlinski

Phonon ver. 5.10

<http://www.ComputingForMaterials.com/>

A new PHONON version 5.10 (32bits), for Windows and Linux.

New option:

- entirely new interface for Windows and Linux
- separated ab initio and modelling options
- automatic updating of lattice constants and atomic positions after DFT optimization
- plot of phonons' intensities by adjustable colours
- easier setting of wavevectors and wave numbers in Brillouin zones
- plot parameters set now by User
- any selected phonon mode can be fully animated by PHONON internal routine
- finds low symmetry space group created by commensurate phonon modes
- plots x-rays scattering with atomic form factor and nuclear inelastic scattering
- calculates dielectric constant (phonon part) and infrared absorption
- provides symmetry analysis of LO/TO

Thank you !